

A stabilized displacement-volumetric strain formulation for nearly incompressible and anisotropic materials

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Abstract

The simulation of structural problems involving the deformations of volumetric bodies is of paramount importance in many areas of engineering. Although the use of tetrahedral elements is extremely appealing, tetrahedral discretizations are generally known as very stiff and are hence often avoided in typical simulation workflows. The development of mixed displacement-pressure approaches has allowed to effectively overcome this problem leading to a class of locking-free elements which can effectively compete with hexahedral discretizations while retaining obvious advantages in the mesh generation step. Despite such advantages the adoption of the technology within commercial codes is not yet pervasive. This can be attributed to two different reasons: the difficulty in making use of standard constitutive libraries and the implied continuity of the pressure, which makes questionable the application of the method in the context of multi-material problems. Current paper proposes the adoption of the volumetric strain instead of the pressure as nodal value. Such choice leads naturally to the definition of a modified strain making straightforward the use of standard strain-driven constitutive laws. At the same time, the continuity of the volumetric strain across multimaterial interfaces can be naturally accepted as a sort of kinematic constraint (stresses can still remain discontinuous across material interfaces). The new element also opens the door to the use of anisotropic constitutive laws, which are typically problematic in the context of mixed elements.

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1. Motivation

The development of mixed Q_1/Q_1 (multi-linear/multi-linear) and P_1/P_1 (linear/linear) displacement-pressure approaches [1] has represented a milestone in the finite element (FE) technology, opening for the first time the possibility of improving the accuracy of low order meshes while guaranteeing a lock-free behaviour at the nearly-incompressible limit. The key idea of displacement-pressure (\mathbf{u} - p) approaches is to split the constitutive response into its deviatoric and volumetric parts. The deviatoric part of the strain is then recovered from the displacement field and fed to the constitutive law which then returns the corresponding deviatoric stress. The volumetric part on the other hand is obtained in terms of the nodal pressure field. Even though this approach effectively solves any volumetric locking issue, it also implies that the total strain is never explicitly computed (in the FE implementation, only deviatoric strain and pressure are available at the Gauss points). The practical downside of this situation is that one cannot make use of standard strain-driven constitutive laws, which represents a practical blocker in the context of commercial codes which need to leverage large material libraries. The proposed approach sidesteps this difficulty by *choosing as primal variable the volumetric strain ε^v instead of the pressure*. In the implementation, the difference is that the total strain can be recovered at the Gauss point level by simply summing the deviatoric part (obtained as before in terms of the displacement gradient) and the volumetric one obtained by interpolation of ε^v . The use of standard constitutive models becomes thus straightforward, effectively resolving the problem described.

A second well known difficulty, which is intrinsic to the use of equal-order, mixed, displacement-pressure fields, is that the pressure is treated as a continuous FE variable. This becomes problematic when multiple materials need to be considered within the domain, since in presence of pressure discontinuities continuous approximations typically manifest unwanted oscillations. Although this can be remedied for example by doubling the pressure degrees of freedom at the interface, [2], such approach is normally inconvenient when more than two materials are present. The use of a continuous discretization for ε^v on the contrary does not prevent discontinuous pressures to rise across the material interface, thus effectively sidestepping such difficulty.

35 Interestingly, for isotropic linear constitutive relations the proposed for-
36 mulation can be understood simply as a change of variable with respect to
37 displacement-pressure approaches. When considered in this context, the $\mathbf{u}-\varepsilon^v$
38 formulation inherits all the stability properties of the original $\mathbf{u}-p$ approach
39 (see e.g. [3] for a recent discussion).

40 We shall also remark that the use of displacement-strain (total strain)
41 formulations has been proposed previously in [4] as an alternative to the
42 displacement-stress approach, also described in [4]. An enhanced three field
43 formulation (displacement-strain-pressure) $\mathbf{u}-\boldsymbol{\varepsilon}-p$ has been not long ago pro-
44 posed in [5]. To the best of our knowledge however this is the first time
45 where a $\mathbf{u}-\varepsilon^v$ formulation is discussed in detail. To this end, the paper is
46 structured as follows: a mixed displacement-volumetric strain formulation
47 for small strain elasticity is derived as a special case of the displacement-
48 strain formulation in Section 2, where the problem is set at the continuous
49 level and a FE discretisation is proposed. The case of anisotropic materials is
50 studied in Section 3, retrofitting the original formulation to allow the solution
51 of anisotropic problems. This is accomplished by a redefinition of the mod-
52 ified volumetric strain which accounts for the anisotropic behaviour of the
53 material. The article is concluded by a set of convergence tests in Section 4,
54 performed both in the isotropic and anisotropic cases, and by a number of
55 test examples proving the performance of the proposed formulation. Finally,
56 the last section collects the conclusion and further work lines of the paper.

57 The $\mathbf{u}-\varepsilon^v$ formulation we propose is implemented and ready to use within
58 the open source Kratos Multiphysics framework [6, 7].

59 **2. Formulation**

60 *2.1. Governing equations*

61 The essence of the proposed formulation is to modify the (small) strain
62 definition to avoid volumetric locking. This is accomplished by employing
63 a mixed formulation in which the volumetric strain ε^v is considered as an
64 unknown, and interpolated as such when the problem is approximated using
65 finite elements. The key idea is that the standard deviatoric-isochoric split-
66 ting is performed at the strain level. The deviatoric part is then computed
67 in terms of the displacements while the isochoric one is expressed in terms

68 of ε^v . This is expressed in symbols as

$$\boldsymbol{\varepsilon}(\mathbf{x}) = \underbrace{\nabla^s \mathbf{u} - \frac{1}{\alpha} \nabla \cdot \mathbf{u} \mathbf{I}}_{\boldsymbol{\varepsilon}_{\text{dev}}} + \underbrace{\frac{1}{\alpha} \varepsilon^v \mathbf{I}}_{\boldsymbol{\varepsilon}_{\text{iso}}} \quad (1)$$

69 where \mathbf{I} is the identity matrix. The coefficient α is taken here as $\alpha = 3$ in the
 70 3D case and $\alpha = 2$ in the 2D one. This choice implies that in 2D cases the
 71 “volumetric” strain should be understood *as a measure of the area change*
 72 *in the plane* rather than a measure of the real volume change.

Once the strain splitting is defined, the equilibrium problem can be written as

$$-\nabla \cdot \boldsymbol{\sigma}(\boldsymbol{\varepsilon}) = \mathbf{f} \quad (2a)$$

$$\nabla \cdot \mathbf{u} - \varepsilon^v = 0 \quad (2b)$$

73 where the first equation is the classical equilibrium condition and the second
 74 expresses the kinematic relation between the volume variation and the displacement field, which is exact in the continuum for the small deformation
 75 case.
 76

77 No assumption is made up to this point about the constitutive behaviour
 78 other than $\boldsymbol{\sigma} = \boldsymbol{\sigma}(\boldsymbol{\varepsilon})$. Likewise, the introduction of the volumetric strain
 79 as a variable can be done both for stationary and time dependent problems,
 80 although we will restrict ourselves to the former in this paper.

81 Furthermore, we note that Eq. 2b can be written in incremental form as

$$\nabla \cdot \Delta \mathbf{u} - \Delta \varepsilon^v = 0 \quad (3)$$

82 $\Delta(\cdot)$ denoting an increment; this choice is completely equivalent to Eq. 2b in
 83 the linear case but has some practical advantages in the application of initial
 84 conditions or initial guesses of iterative schemes.

85 2.2. Variational approach

86 Obtaining a symmetric variational form for the problem described in Eqs.
 87 2a and 2b is not obvious. Our approach for doing so is to begin by considering
 88 the mixed displacement-strain form, described in [4] or for the explicit case
 89 in [8, 9].

90 *2.2.1. Standard \mathbf{u} - $\boldsymbol{\varepsilon}$ formulation*

Let us start considering the differential form of the $\mathbf{u} - \boldsymbol{\varepsilon}$ formulation, which reads:

$$\begin{aligned} -\nabla \cdot \mathbb{C} : \boldsymbol{\varepsilon} &= \mathbf{f} \\ \mathbb{C} : \boldsymbol{\varepsilon} - \mathbb{C} : \nabla^s \mathbf{u} &= \mathbf{0} \end{aligned}$$

91 where \mathbb{C} is the constitutive tensor and \mathbf{f} the vector of external body forces.
 92 To simplify the exposition, let us consider homogeneous Dirichlet boundary
 93 conditions $\mathbf{u} = \mathbf{0}$ on the whole boundary $\partial\Omega$ of the domain Ω where the
 94 problem is posed.

Let $\boldsymbol{\delta}_{\mathbf{u}}$ be the displacement test function (vanishing on $\partial\Omega$) and $\boldsymbol{\delta}_{\boldsymbol{\varepsilon}}$ the strain test function. The weak form of the problem consists of finding \mathbf{u} and $\boldsymbol{\varepsilon}$ in the appropriate spaces such that

$$\int_{\Omega} \nabla^s \boldsymbol{\delta}_{\mathbf{u}} : \mathbb{C} : \boldsymbol{\varepsilon} = \int_{\Omega} \boldsymbol{\delta}_{\mathbf{u}} \cdot \mathbf{f} \quad (4a)$$

$$- \int_{\Omega} \boldsymbol{\delta}_{\boldsymbol{\varepsilon}} : \mathbb{C} : (\boldsymbol{\varepsilon} - \nabla^s \mathbf{u}) = 0 \quad (4b)$$

for all test functions $\boldsymbol{\delta}_{\mathbf{u}}$ and $\boldsymbol{\delta}_{\boldsymbol{\varepsilon}}$. The problem can also be written in the form

$$B_{\mathbf{u}\boldsymbol{\varepsilon}}(\mathbf{u}, \boldsymbol{\varepsilon}; \boldsymbol{\delta}_{\mathbf{u}}, \boldsymbol{\delta}_{\boldsymbol{\varepsilon}}) := \int_{\Omega} \nabla^s \boldsymbol{\delta}_{\mathbf{u}} : \mathbb{C} : \boldsymbol{\varepsilon} - \int_{\Omega} \boldsymbol{\delta}_{\boldsymbol{\varepsilon}} : \mathbb{C} : (\boldsymbol{\varepsilon} - \nabla^s \mathbf{u}) = \int_{\Omega} \boldsymbol{\delta}_{\mathbf{u}} \cdot \mathbf{f} \quad (5)$$

It is observed that the bilinear form $B_{\mathbf{u}\boldsymbol{\varepsilon}}$ is semi-definite:

$$B_{\mathbf{u}\boldsymbol{\varepsilon}}(\mathbf{u}, \boldsymbol{\varepsilon}; \mathbf{u}, -\boldsymbol{\varepsilon}) = \int_{\Omega} \boldsymbol{\varepsilon} : \mathbb{C} : \boldsymbol{\varepsilon}$$

95 From this one can easily get a stability estimate for the strain, but not for the
 96 displacement. An inf-sup condition is required to bound it in the continuous
 97 case which needs to be inherited by the FE interpolation, unless a stabilised
 98 FE method is employed. A similar comment applies to the formulation to
 99 be proposed later.

If we introduce the functional

$$\mathcal{E}_{\mathbf{u}\boldsymbol{\varepsilon}}(\mathbf{u}, \boldsymbol{\varepsilon}) = \frac{1}{2} \int_{\Omega} (\boldsymbol{\varepsilon} - \nabla^s \mathbf{u}) : \mathbb{C} : (\boldsymbol{\varepsilon} - \nabla^s \mathbf{u}) - \frac{1}{2} \int_{\Omega} \nabla^s \mathbf{u} : \mathbb{C} : \nabla^s \mathbf{u} + \int_{\Omega} \mathbf{u} \cdot \mathbf{f}$$

100 it is easily seen that Eqs. 4 are precisely its stationary conditions. The way
 101 we have written $\mathcal{E}_{\mathbf{u}\boldsymbol{\varepsilon}}$ is intended to motivate the following formulation.

102 *2.2.2. \mathbf{u} - ε^v formulation*

Our proposal is to start from the variational form of the \mathbf{u} - $\boldsymbol{\varepsilon}$ formulation and to substitute the strain formula $\boldsymbol{\varepsilon} := \nabla^s \mathbf{u} - \frac{1}{\alpha} \nabla \cdot \mathbf{u} \mathbf{I} + \frac{1}{\alpha} \varepsilon^v \mathbf{I}$ into it. Thus, let us consider the functional

$$\begin{aligned} \mathcal{E}_{\mathbf{u}\varepsilon^v}(\mathbf{u}, \varepsilon^v) &= \frac{1}{2} \frac{1}{\alpha^2} \int_{\Omega} (\varepsilon^v - \nabla \cdot \mathbf{u}) \mathbf{I} : \mathbb{C} : \mathbf{I} (\varepsilon^v - \nabla \cdot \mathbf{u}) \\ &\quad - \frac{1}{2} \int_{\Omega} \nabla^s \mathbf{u} : \mathbb{C} : \nabla^s \mathbf{u} + \int_{\Omega} \mathbf{u} \cdot \mathbf{f} \end{aligned} \quad (6)$$

Let us call

$$\kappa := \frac{1}{\alpha^2} \mathbf{I} : \mathbb{C} : \mathbf{I} \quad (7)$$

which coincides with the volumetric modulus for isotropic materials. It allows us to write the stationary conditions of the functional in Eq. 6 as

$$\begin{aligned} B_{\mathbf{u}\varepsilon^v}(\mathbf{u}, \varepsilon^v; \boldsymbol{\delta}_{\mathbf{u}}, \delta_{\varepsilon^v}) &:= \int_{\Omega} (\delta_{\varepsilon^v} - \nabla \cdot \boldsymbol{\delta}_{\mathbf{u}}) \kappa (\varepsilon^v - \nabla \cdot \mathbf{u}) \\ &\quad - \int_{\Omega} \nabla^s \boldsymbol{\delta}_{\mathbf{u}} : \mathbb{C} : \nabla^s \mathbf{u} = - \int_{\Omega} \boldsymbol{\delta}_{\mathbf{u}} \cdot \mathbf{f} \end{aligned} \quad (8)$$

for all test functions $\boldsymbol{\delta}_{\mathbf{u}}, \delta_{\varepsilon^v}$, where $B_{\mathbf{u}\varepsilon^v}$ is the counterpart of the bilinear form $B_{\mathbf{u}\boldsymbol{\varepsilon}}$ in Eq. 5 for the formulation we propose. Problem 8 can also be split as

$$\int_{\Omega} \nabla^s \boldsymbol{\delta}_{\mathbf{u}} : \mathbb{C} : \nabla^s \mathbf{u} + \int_{\Omega} \nabla \cdot \boldsymbol{\delta}_{\mathbf{u}} \kappa (\varepsilon^v - \nabla \cdot \mathbf{u}) = \int_{\Omega} \boldsymbol{\delta}_{\mathbf{u}} \cdot \mathbf{f} \quad (9a)$$

$$\int_{\Omega} \delta_{\varepsilon^v} \kappa (\varepsilon^v - \nabla \cdot \mathbf{u}) = 0 \quad (9b)$$

for all test functions $\boldsymbol{\delta}_{\mathbf{u}}, \delta_{\varepsilon^v}$, which is the counterpart of Problem 4 for the \mathbf{u} - ε^v formulation. The strong (differential) form of these equations (for κ constant) is:

$$-\nabla \cdot \mathbb{C} : \nabla^s \mathbf{u} - \kappa \nabla (\varepsilon^v - \nabla \cdot \mathbf{u}) = \mathbf{f} \quad (10a)$$

$$\varepsilon^v - \nabla \cdot \mathbf{u} = 0 \quad (10b)$$

103 Recall that zero Dirichlet conditions have been assumed throughout.

Remark 1. In the case of arbitrary stress-strain relations, Problem 9 can be modified by replacing $\mathbb{C} : \nabla^s \mathbf{u}$ by the stress $\boldsymbol{\sigma}(\boldsymbol{\varepsilon})$ and introducing a scaling physical parameter $\tilde{\kappa}$ (with the same units as κ), so that the variational form of the problem would be

$$\int_{\Omega} \nabla^s \boldsymbol{\delta}_{\mathbf{u}} : \boldsymbol{\sigma}(\boldsymbol{\varepsilon}) + \int_{\Omega} \nabla \cdot \boldsymbol{\delta}_{\mathbf{u}} \tilde{\kappa}(\varepsilon^v - \nabla \cdot \mathbf{u}) = \int_{\Omega} \boldsymbol{\delta}_{\mathbf{u}} \cdot \mathbf{f} \quad (11a)$$

$$\int_{\Omega} \delta_{\varepsilon^v} \tilde{\kappa}(\varepsilon^v - \nabla \cdot \mathbf{u}) = 0 \quad (11b)$$

104 for all test functions $\boldsymbol{\delta}_{\mathbf{u}}$ and δ_{ε^v} . \square

Remark 2. Even though no assumption on \mathbb{C} has been stated to obtain Problem 9, we will use it for isotropic materials; the way we deal with anisotropic cases is explained in Section 3. Consider then an isotropic material, and let us introduce Π_{dev} as the projection of second order tensors onto their deviatoric component. We may write Eq. 10a as

$$-\nabla \cdot \Pi_{\text{dev}}(\mathbb{C} : \nabla^s \mathbf{u}) - \frac{1}{\alpha} \nabla \cdot (\nabla \cdot \mathbf{u} \mathbb{C} : \mathbf{I}) - \kappa \nabla(\varepsilon^v - \nabla \cdot \mathbf{u}) = \mathbf{f} \quad (12)$$

For isotropic materials

$$\frac{1}{\alpha} \nabla \cdot (\nabla \cdot \mathbf{u} \mathbb{C} : \mathbf{I}) = \kappa \nabla(\nabla \cdot \mathbf{u})$$

and then Eq. 12 can be written as

$$-\nabla \cdot \Pi_{\text{dev}}(\mathbb{C} : \nabla^s \mathbf{u}) - \kappa \nabla \varepsilon^v = \mathbf{f}$$

105 The change of variable $p = \kappa \varepsilon^v$ yields the classical \mathbf{u} - p formulation of linear
106 elasticity, which would allow us to deal with purely incompressible materials,
107 i.e., $\kappa = \infty$. In this case, Eq. 10b would be $\nabla \cdot \mathbf{u} = 0$. \square

108 **Remark 3.** In line with the previous remark, let us notice that for anisotropic
109 materials the incompressibility condition $\nabla \cdot \mathbf{u} = 0$ is *not* implied by any limit
110 value of a physical property, as in the isotropic case, but by different condi-
111 tions that relate the physical properties of an anisotropic material (see for
112 example [10, 11]). \square

113 *2.3. Variational Multi-Scale stabilisation*

Let us consider the continuous problem given in Eq. 8. The bilinear form of the problem satisfies:

$$B_{\mathbf{u}\varepsilon^v}(\mathbf{u}, \varepsilon^v; -\mathbf{u}, \varepsilon^v) = \int_{\Omega} \kappa(\varepsilon^v)^2 - \int_{\Omega} \kappa(\nabla \cdot \mathbf{u})^2 + \int_{\Omega} \nabla^s \mathbf{u} : \mathbb{C} : \nabla^s \mathbf{u} \quad (13)$$

114 For isotropic materials, the second term is precisely the volumetric compo-
 115 nent of the third one and, since the deviatoric and volumetric components
 116 of a tensor are orthogonal, we are left with only the deviatoric part. In the
 117 case of anisotropic or nonlinear materials, the scaling coefficient $\tilde{\kappa}$ should
 118 be chosen such that the second term could be absorbed by the third one.
 119 In any case, it is observed that this expression provides control on the de-
 120 viatoric part of $\nabla^s \mathbf{u}$ and on ε^v . Thus, we miss control on the volumetric
 121 part of $\nabla^s \mathbf{u}$, which at the continuous level can be obtained from an inf-sup
 122 condition from the control on ε^v . However, if we use the standard Galerkin
 123 FE discretisation, this inf-sup condition will not necessarily hold. Moreover,
 124 since derivatives of ε^v do not appear in Eq. 13, there is no guarantee to have
 125 them bounded, and the FE approximation to this variable may display node-
 126 to-node oscillations. This effect is particularly important in materials close
 127 to incompressible, case in which $\varepsilon^v \rightarrow 0$ and, even if $\kappa \rightarrow \infty$, $\kappa(\varepsilon^v)^2 \rightarrow 0$
 128 (since $\kappa\varepsilon^v$ must remain bounded).

129 In our numerical experiments we have observed that the Galerkin approx-
 130 imation to the problem in Eq. 8 is polluted by strong numerical oscillations.
 131 This is why we present now a stabilised FE formulation based on the Varia-
 132 tional Multi-Scale (VMS) concept [12, 13].

133 Let us consider the domain Ω discretised in a partition $\{\Omega^e\}$ of elements of
 134 characteristic size h , index e ranging from 1 to the total number of elements.
 135 From this, we may construct the interpolating spaces for \mathbf{u} and ε^v ; standard
 136 continuous Lagrangian interpolations will be assumed for both variables. FE
 137 functions will be identified with the subscript h .

The VMS method is based on the separation of the unknown fields, in
 this case the displacement \mathbf{u} and the volumetric strain ε^v , in two scales. On
 one hand we have the scale which can be represented by the FE solution, \mathbf{u}_h
 and ε_h^v . On the other hand we have the so called sub-scales, which represent
 the part of the solution that cannot be captured by the FE mesh and needs
 to be modelled. The sub-scales are denoted with the subindex s , as \mathbf{u}_s and
 ε_s^v . We thus have the decomposition

$$\mathbf{u} = \mathbf{u}_h + \mathbf{u}_s \quad (14a)$$

$$\varepsilon^v = \varepsilon_h^v + \varepsilon_s^v \quad (14b)$$

138 A similar splitting holds for the test functions, yielding the equation in the
 139 FE space and in the space of sub-scales. The idea is now to insert these
 140 splittings into the variational form of the problem, integrate by parts terms
 141 involving derivatives of the sub-scales and then give an approximation for
 142 them (not for their derivatives).

Let us consider then Problem 9 taking the test functions in the corresponding FE spaces, using the splittings in Eqs. 14 and integrating by parts within each element. The result is:

$$\begin{aligned} & \int_{\Omega} \nabla^s \boldsymbol{\delta}_{\mathbf{u}_h} : \mathbb{C} : \nabla^s \mathbf{u}_h - \sum_e \int_{\Omega^e} \mathbf{u}_s \cdot \nabla \cdot \mathbb{C} : \nabla^s \boldsymbol{\delta}_{\mathbf{u}_h} \\ & + \int_{\Omega} \nabla \cdot \boldsymbol{\delta}_{\mathbf{u}_h} \kappa (\varepsilon_h^v + \varepsilon_s^v - \nabla \cdot \mathbf{u}_h) + \sum_e \int_{\Omega^e} \mathbf{u}_s \cdot \kappa \nabla \nabla \cdot \boldsymbol{\delta}_{\mathbf{u}_h} \\ & = \int_{\Omega} \boldsymbol{\delta}_{\mathbf{u}_h} \cdot \mathbf{f} \end{aligned} \quad (15a)$$

$$\int_{\Omega} \delta_{\varepsilon_h^v} \kappa (\varepsilon_h^v + \varepsilon_s^v - \nabla \cdot \mathbf{u}_h) + \sum_e \int_{\Omega^e} \mathbf{u}_s \cdot \kappa \nabla \delta_{\varepsilon_h^v} = 0 \quad (15b)$$

where sub-scales on the element boundaries have been discarded, although this assumption can be relaxed [14]. Writing Eqs. 15 together we have:

$$\begin{aligned} & \int_{\Omega} \nabla^s \boldsymbol{\delta}_{\mathbf{u}_h} : \mathbb{C} : \nabla^s \mathbf{u}_h + \int_{\Omega} (\delta_{\varepsilon_h^v} + \nabla \cdot \boldsymbol{\delta}_{\mathbf{u}_h}) \kappa (\varepsilon_h^v - \nabla \cdot \mathbf{u}_h) \\ & + \sum_e \int_{\Omega^e} \mathbf{u}_s \cdot [-\nabla \cdot \mathbb{C} : \nabla^s \boldsymbol{\delta}_{\mathbf{u}_h} + \kappa \nabla (\varepsilon_h^v + \nabla \cdot \boldsymbol{\delta}_{\mathbf{u}_h})] \\ & + \sum_e \int_{\Omega^e} \varepsilon_s^v \kappa (\delta_{\varepsilon_h^v} + \nabla \cdot \boldsymbol{\delta}_{\mathbf{u}_h}) = \int_{\Omega} \boldsymbol{\delta}_{\mathbf{u}_h} \cdot \mathbf{f} \end{aligned} \quad (16)$$

The model is completed by the choice of an approximation for the sub-scales. The counterpart of Eq. 16 with the test functions taken in the space of sub-scales would lead to an equation projected onto this space stating that the differential operator of the problem is equal to the residual of the FE scales. This operator applied to the sub-scales can then be approximated by a diagonal matrix using different arguments (see [13] for a review and details). In view of the equations to be solved 10, the final result is:

$$\mathbf{u}_s = \tau_1 P_s [\mathbf{f} + \nabla \cdot \mathbb{C} : \nabla^s \mathbf{u}_h + \kappa \nabla (\varepsilon_h^v - \nabla \cdot \mathbf{u}_h)] \quad (17a)$$

$$\varepsilon_s^v = \tau_2 P_s [\nabla \cdot \mathbf{u}_h - \varepsilon_h^v] \quad (17b)$$

143 where τ_1 and τ_2 are the stabilisation parameters, given below, and P_s is the
 144 projection onto the space of sub-scales, either of \mathbf{u}_s or of ε_s^v .

Inserting the sub-scales given by Eqs. 17 into Eq. 16 we finally obtain the stabilised FE method we propose, which consists in finding \mathbf{u}_h and ε_h^v such that

$$\begin{aligned} & B_{\mathbf{u}\varepsilon^v, \text{stab}}(\mathbf{u}_h, \varepsilon_h^v; \boldsymbol{\delta}_{\mathbf{u}_h}, \delta_{\varepsilon_h^v}) \\ & := \int_{\Omega} \nabla^s \boldsymbol{\delta}_{\mathbf{u}_h} : \mathbb{C} : \nabla^s \mathbf{u}_h + \int_{\Omega} (\delta_{\varepsilon_h^v} + \nabla \cdot \boldsymbol{\delta}_{\mathbf{u}_h}) \kappa (\varepsilon_h^v - \nabla \cdot \mathbf{u}_h) \\ & \quad + \sum_e \int_{\Omega^e} \tau_1 P_s [\nabla \cdot \mathbb{C} : \nabla^s \mathbf{u}_h + \kappa \nabla (\varepsilon_h^v - \nabla \cdot \mathbf{u}_h)] \\ & \quad \quad \cdot [-\nabla \cdot \mathbb{C} : \nabla^s \boldsymbol{\delta}_{\mathbf{u}_h} + \kappa \nabla (\delta_{\varepsilon_h^v} + \nabla \cdot \boldsymbol{\delta}_{\mathbf{u}_h})] \\ & \quad + \sum_e \int_{\Omega^e} \tau_2 P_s (\nabla \cdot \mathbf{u}_h - \varepsilon_h^v) \kappa (\delta_{\varepsilon_h^v} + \nabla \cdot \boldsymbol{\delta}_{\mathbf{u}_h}) \\ & = \int_{\Omega} \boldsymbol{\delta}_{\mathbf{u}_h} \cdot \mathbf{f} - \sum_e \int_{\Omega^e} \tau_1 P_s [\mathbf{f}] \cdot [-\nabla \cdot \mathbb{C} : \nabla^s \boldsymbol{\delta}_{\mathbf{u}_h} + \kappa \nabla (\delta_{\varepsilon_h^v} + \nabla \cdot \boldsymbol{\delta}_{\mathbf{u}_h})] \\ & =: L_{\mathbf{u}\varepsilon^v, \text{stab}}(\boldsymbol{\delta}_{\mathbf{u}_h}, \delta_{\varepsilon_h^v}) \end{aligned} \quad (18)$$

145 for all test functions $\boldsymbol{\delta}_{\mathbf{u}_h}$ and $\delta_{\varepsilon_h^v}$.

146 To complete the definition of the method, we need to define the projection
 147 P_s and the expression of stabilisation parameters. Even though the space
 148 for the sub-scales can be defined in different manners (bubble functions,
 149 approximation to Green's function,...) when arriving to Eq. 17 there are
 150 essentially two options, namely, to take the space of sub-scale as the space
 151 of FE residuals, yielding $P_s = I$ (the identity) or to take it as L^2 orthogonal
 152 to the FE space, case in which P_s is the orthogonal projection to this space.
 153 The second option has theoretical and practical advantages, as reported for
 154 example in [4, 15, 16, 17]. However, here we will consider the most common
 155 option of taking $P_s = I$, leading to classical residual-based stabilised FE
 156 methods. See also [13] for further discussion.

Regarding the stabilisation parameters, they can be determined by scaling arguments or by assuming that the sub-scales are bubble functions. In either case, the result is that they should behave as

$$\tau_1 = c_1 \frac{h^2}{G}, \quad \tau_2 = c_2 \frac{G}{G + \kappa} \quad (19)$$

157 where G an equivalent effective shear modulus and c_1 and c_2 are algorithmic
 158 constants, that in the case of linear elements we take as $c_1 = 2$, $c_2 = 4$. Let
 159 us remark that the definition of “equivalent effective shear modulus” is not
 160 univocal in the context of anisotropic materials. We defer the discussion on
 161 the exact definition of such term to the following sections.

The formulation we propose is given by Eq. 18 with $P_s = I$ and τ_1 and τ_2
 given in Eq. 19. Let us consider the case of linear elements, in which second
 derivatives inside the elements are zero. Eq. 18 can then be arranged to give

$$\begin{aligned}
 & B_{\mathbf{u}\varepsilon^v, \text{stab, lin}}(\mathbf{u}_h, \varepsilon_h^v; \boldsymbol{\delta}_{\mathbf{u}_h}, \delta_{\varepsilon_h^v}) \\
 & := \int_{\Omega} \nabla^s \boldsymbol{\delta}_{\mathbf{u}_h} : \mathbb{C} : \nabla^s \mathbf{u}_h + \int_{\Omega} (1 - \tau_2) (\delta_{\varepsilon_h^v} + \nabla \cdot \boldsymbol{\delta}_{\mathbf{u}_h}) \kappa (\varepsilon_h^v - \nabla \cdot \mathbf{u}_h) \\
 & \quad + \int_{\Omega} \tau_1 \kappa^2 \nabla \delta_{\varepsilon_h^v} \cdot \nabla \varepsilon_h^v = \int_{\Omega} \boldsymbol{\delta}_{\mathbf{u}_h} \cdot \mathbf{f} - \int_{\Omega} \tau_1 \mathbf{f} \cdot \kappa \nabla \delta_{\varepsilon_h^v} \tag{20}
 \end{aligned}$$

Remark 4. Even though it is not the purpose of this paper to analyse the
 stability and convergence properties of the method in detail, the simplified
 problem in Eq. 20 allows us to understand the effect on stability of τ_1 and
 τ_2 . Assuming these for simplicity to be constant, we have that

$$\begin{aligned}
 B_{\mathbf{u}\varepsilon^v, \text{stab, lin}}(\mathbf{u}_h, \varepsilon_h^v; \mathbf{u}_h, \varepsilon_h^v) &= \|\mathbb{C}^{1/2} : \nabla^s \mathbf{u}_h\|^2 + (1 - \tau_2) \|\kappa^{1/2} \varepsilon_h^v\|^2 \\
 &\quad - (1 - \tau_2) \|\kappa^{1/2} \nabla \cdot \mathbf{u}_h\|^2 + \tau_1 \|\kappa \nabla \delta_{\varepsilon_h^v}\|^2
 \end{aligned}$$

162 where $\mathbb{C}^{1/2}$ is the square root of the positive-definite tensor \mathbb{C} and $\|\cdot\|$ is the
 163 L^2 norm in Ω . From this expression we observe that

- 164 • τ_2 reduces the (positive) L^2 control on ε_h^v .
- 165 • τ_2 reduces the subtracting L^2 norm of $\nabla \cdot \mathbf{u}_h$.
- 166 • τ_1 provides control on the derivatives of ε_h^v .

167 It is observed that the crucial parameter from the numerical point of view is
 168 τ_1 and that we need to ensure that $\tau_2 < 1$. □

Remark 5. In order to be able to use generic materials we may proceed as
 indicated in Remark 1. If $\tilde{\kappa}$ is an adequate scaling physical parameter, the
 problem to be solved for a general constitutive law $\boldsymbol{\sigma} = \boldsymbol{\sigma}(\boldsymbol{\varepsilon})$ is

$$\int_{\Omega} \nabla^s \boldsymbol{\delta}_{\mathbf{u}_h} : \boldsymbol{\sigma}(\boldsymbol{\varepsilon}_h) + \int_{\Omega} (1 - \tau_2) (\delta_{\varepsilon_h^v} + \nabla \cdot \boldsymbol{\delta}_{\mathbf{u}_h}) \tilde{\kappa} (\varepsilon_h^v - \nabla \cdot \mathbf{u}_h)$$

$$+ \int_{\Omega} \tau_1 \tilde{\kappa}^2 \nabla \delta_{\varepsilon_h^v} \cdot \nabla \varepsilon_h^v = \int_{\Omega} \boldsymbol{\delta}_{\mathbf{u}_h} \cdot \mathbf{f} - \int_{\Omega} \tau_1 \mathbf{f} \cdot \tilde{\kappa} \nabla \delta_{\varepsilon_h^v} \quad (21)$$

169 We remark here that the stabilization factor τ_2 does not enter in the definition
170 of the FE strain $\boldsymbol{\varepsilon}_h$, and is hence not employed in the calculation of the stress.

The formulation given by Eq. 21 reduces to the linear one when one uses the strain $\boldsymbol{\varepsilon}_h = \nabla^s \mathbf{u}_h$ in the constitutive law. A different choice is to use instead the modified strain (following the, admittedly heuristic, rationale that such strain is "better" at the Gauss point level)

$$\boldsymbol{\varepsilon}_h := \nabla^s \mathbf{u}_h - \frac{1}{\alpha} \nabla \cdot \mathbf{u}_h \mathbf{I} + \frac{1}{\alpha} \varepsilon_h^v \mathbf{I} \quad (22)$$

Should this be the case, the first term in Eq. 21 becomes

$$\int_{\Omega} \nabla^s \boldsymbol{\delta}_{\mathbf{u}_h} : \left[\boldsymbol{\sigma}(\boldsymbol{\varepsilon}_h) - \mathbb{C} : \left(-\frac{1}{\alpha} \nabla \cdot \mathbf{u}_h \mathbf{I} + \frac{1}{\alpha} \varepsilon_h^v \mathbf{I} \right) \right] \quad (23)$$

171 where $\mathbb{C} := \left. \frac{\partial \boldsymbol{\sigma}}{\partial \boldsymbol{\varepsilon}_h} \right|_{\boldsymbol{\varepsilon}_h}$ should be interpreted as the tangent constitutive tensor
172 of the constitutive law.

173 As we will show later, the tangent matrix of a Newton-Raphson linearisation
174 of the problem in Eq. 21 using $\boldsymbol{\varepsilon}_h = \nabla^s \mathbf{u}_h$ is identical to the one obtained
175 using the modification in Eq. 23 with the strain given by Eq. 22, although
176 the residual would of course be different. In our numerical examples we have
177 employed the modification in Eq. 23, although similar results are expected
178 when this modification is not done. \square

179 2.4. Finite Element Implementation—Isotropic case

180 A number of, rather standard, definitions are useful to write the FE
181 discretisation of the proposed discrete variational problem 21. Let N_I be the
182 standard (Lagrangian) shape function of node I of the FE mesh, and x, y, z
183 the Cartesian coordinates, and let us introduce the following arrays, whose
184 definition depends on the number of space dimensions:

$$\mathbf{B}_I = \begin{pmatrix} \frac{\partial N_I}{\partial x} & 0 & 0 \\ 0 & \frac{\partial N_I}{\partial y} & 0 \\ 0 & 0 & \frac{\partial N_I}{\partial x} \\ \frac{\partial N_I}{\partial y} & \frac{\partial N_I}{\partial x} & 0 \\ 0 & \frac{\partial N_I}{\partial z} & \frac{\partial N_I}{\partial y} \\ \frac{\partial N_I}{\partial z} & 0 & \frac{\partial N_I}{\partial x} \end{pmatrix} (3D), \quad \mathbf{B}_I = \begin{pmatrix} \frac{\partial N_I}{\partial x} & 0 \\ 0 & \frac{\partial N_I}{\partial y} \\ \frac{\partial N_I}{\partial y} & \frac{\partial N_I}{\partial x} \end{pmatrix} (2D) \quad (24)$$

185

$$\mathbf{m} := \begin{pmatrix} 1 \\ 1 \\ 1 \\ 0 \\ 0 \\ 0 \end{pmatrix} (3D), \quad \mathbf{m} := \begin{pmatrix} 1 \\ 1 \\ 0 \end{pmatrix} (2D) \quad (25)$$

$$\mathbf{G}_I := \begin{pmatrix} \frac{\partial N_I}{\partial x} \\ \frac{\partial N_I}{\partial y} \\ \frac{\partial N_I}{\partial z} \end{pmatrix} (3D), \quad \mathbf{G}_I := \begin{pmatrix} \frac{\partial N_I}{\partial x} \\ \frac{\partial N_I}{\partial y} \end{pmatrix} (2D) \quad (26)$$

186

$$\mathbf{P} := \mathbf{I} - \frac{1}{\alpha} \mathbf{m} \mathbf{m}^t \quad (27)$$

187

$$\kappa := \frac{\mathbf{m}^t \mathbf{C} \mathbf{m}}{\alpha^2} \quad (28)$$

188 where \mathbf{C} is the Voigt representation of the tangent constitutive tensor $\mathbb{C} :=$

189
$$\left. \frac{\partial \boldsymbol{\sigma}}{\partial \boldsymbol{\varepsilon}_h} \right|_{\boldsymbol{\varepsilon}_h}.$$

190 The FE residual assumes finally a slightly different form depending on the
191 choice of $\boldsymbol{\varepsilon}_h$ (see Remark 5). Namely if we choose $\boldsymbol{\varepsilon}_h := \nabla^s \mathbf{u}_h - \frac{1}{\alpha} \nabla \cdot \mathbf{u}_h \mathbf{I} + \frac{1}{\alpha} \boldsymbol{\varepsilon}_h^v \mathbf{I}$
192 (option we followed in our implementation) the residual is

$$\mathbf{R}_I := \begin{pmatrix} N_I \mathbf{f} - \mathbf{B}_I^t \boldsymbol{\sigma}(\boldsymbol{\varepsilon}_h) + \frac{1}{\alpha} \mathbf{B}_I^t \mathbf{C} \mathbf{m} (N_J \boldsymbol{\varepsilon}_{h,J}^v - \mathbf{G}_J^t \mathbf{u}_{h,J}) - (1 - \tau_2) \kappa \mathbf{G}_I (N_J \boldsymbol{\varepsilon}_{h,J}^v - \mathbf{G}_J^t \mathbf{u}_{h,J}) \\ (1 - \tau_2) \kappa N_I (N_J \boldsymbol{\varepsilon}_{h,J}^v - \mathbf{G}_J^t \mathbf{u}_{h,J}) + \kappa^2 \mathbf{G}_I^t \tau_1 \mathbf{G}_J \boldsymbol{\varepsilon}_{h,J}^v - \kappa \mathbf{G}_I^t \tau_1 \mathbf{f} \end{pmatrix} \quad (29)$$

193 If instead we choose $\boldsymbol{\varepsilon}_h := \nabla^s \mathbf{u}_h$ the residual simplifies to

$$\mathbf{R}_I := \begin{pmatrix} N_I \mathbf{f} - \mathbf{B}_I^t \boldsymbol{\sigma}(\boldsymbol{\varepsilon}_h) - (1 - \tau_2) \kappa \mathbf{G}_I (N_J \boldsymbol{\varepsilon}_{h,J}^v - \mathbf{G}_J^t \mathbf{u}_{h,J}) \\ (1 - \tau_2) \kappa N_I (N_J \boldsymbol{\varepsilon}_{h,J}^v - \mathbf{G}_J^t \mathbf{u}_{h,J}) + \kappa^2 \mathbf{G}_I^t \tau_1 \mathbf{G}_J \boldsymbol{\varepsilon}_{h,J}^v - \kappa \mathbf{G}_I^t \tau_1 \mathbf{f} \end{pmatrix} \quad (30)$$

The definition of the discrete problem is completed by the Newton–Raphson linearization. The derivative of the stress term can be computed as

$$\begin{aligned} \mathbf{B}_I^t \frac{\partial \boldsymbol{\sigma}(\boldsymbol{\varepsilon})}{\partial \mathbf{u}_{h,J}} &= \mathbf{B}_I^t \frac{\partial \boldsymbol{\sigma}(\boldsymbol{\varepsilon})}{\partial \boldsymbol{\varepsilon}} \frac{\partial \boldsymbol{\varepsilon}}{\partial \mathbf{u}_{h,J}} = \mathbf{B}_I^t \frac{\partial \boldsymbol{\sigma}(\boldsymbol{\varepsilon})}{\partial \boldsymbol{\varepsilon}} \frac{\partial (\nabla^s \mathbf{u}_h - \frac{1}{\alpha} \nabla \cdot \mathbf{u}_h \mathbf{I} + \frac{1}{\alpha} \boldsymbol{\varepsilon}_h^v \mathbf{I})}{\partial \mathbf{u}_{h,J}} \\ &= \mathbf{B}_I^t \mathbf{C} \mathbf{B}_J - \frac{1}{\alpha} \mathbf{B}_I^t \mathbf{C} \mathbf{m} \mathbf{G}_J^t \end{aligned} \quad (31)$$

and

$$\begin{aligned} \mathbf{B}_I^t \frac{\partial \boldsymbol{\sigma}(\boldsymbol{\varepsilon})}{\partial \varepsilon_{hJ}^v} &= \mathbf{B}_I^t \frac{\partial \boldsymbol{\sigma}(\boldsymbol{\varepsilon})}{\partial \boldsymbol{\varepsilon}} \frac{\partial \boldsymbol{\varepsilon}}{\partial \varepsilon_{hJ}^v} = \mathbf{B}_I^t \frac{\partial \boldsymbol{\sigma}(\boldsymbol{\varepsilon})}{\partial \boldsymbol{\varepsilon}} \frac{\partial (\nabla^s \mathbf{u}_h - \frac{1}{\alpha} \nabla \cdot \mathbf{u}_h \mathbf{I} + \frac{1}{\alpha} \varepsilon_h^v \mathbf{I})}{\partial \varepsilon_{hJ}^v} \\ &= \frac{1}{\alpha} \mathbf{B}_I^t \mathbf{C} \mathbf{m} N_J \end{aligned} \quad (32)$$

This allows to obtain the tangent operator as

$$\begin{aligned} \text{LHS}_{IJ} := & \\ & \begin{pmatrix} \mathbf{B}_I^t \mathbf{C} \mathbf{B}_J - (1 - \tau_2) \kappa \mathbf{G}_I \mathbf{G}_J^t & (1 - \tau_2) \kappa \mathbf{G}_I N_J \\ (1 - \tau_2) \kappa N_I \mathbf{G}_J^t & - (1 - \tau_2) \kappa N_I N_J - \tau_1 \kappa^2 \mathbf{G}_I^t \mathbf{G}_J \end{pmatrix} \end{aligned} \quad (33)$$

194 providing as expected a symmetric tangent (provided that \mathbf{C} is symmetric).

195 **Remark 6.** Note that the same expression of the tangent is obtained inde-
 196 pendently on the definition of $\boldsymbol{\varepsilon}_h$. We observe however that for a non linear
 197 material the actual value of the elasticity tensor (which we recall is defined
 198 as $\mathbb{C} := \left. \frac{\partial \boldsymbol{\sigma}}{\partial \boldsymbol{\varepsilon}_h} \right|_{\boldsymbol{\varepsilon}_h}$) may vary depending on the definition of $\boldsymbol{\varepsilon}_h$, thus finally
 199 resulting in a different stiffness. \square

200 3. Anisotropy

201 The proposed formulation works nicely when the material is approxi-
 202 mately isotropic; however, experimentation with strongly anisotropic mate-
 203 rials shows that instabilities appear in both the volumetric strain and in the
 204 displacement fields. A possibility to address this problem is to try to reduce
 205 the anisotropic case to a “similar” isotropic problem, for which the method
 206 is known to work good. To this end, we observe that any anisotropic ten-
 207 sor \mathbb{C} can be written as $\mathbb{C} = \mathbb{T}^t : \hat{\mathbb{C}} : \mathbb{T}$ where $\hat{\mathbb{C}}$ is an *isotropic* elasticity
 208 tensor. Such property will allow us to propose a slight change in the choice
 209 of our modified volumetric strain. The following subsections detail first the
 210 construction of the “isotropic mapping” and later introduce the proposed
 211 change in the volumetric strain definition.

212 3.1. Constitutive tensor scaling: the closest isotropic tensor

213 The property $\mathbb{C} = \mathbb{T}^t : \hat{\mathbb{C}} : \mathbb{T}$ is easily proved constructively. Let us assume
 214 that \mathbf{C} and $\hat{\mathbf{C}}$ are the Voigt counterparts of \mathbb{C} and $\hat{\mathbb{C}}$. Such matrices are
 215 known to be symmetric and positive definite and hence admit a square root.
 216 Thus if we define $\mathbf{c} := \mathbf{C}^{1/2}$ and $\hat{\mathbf{c}} := \hat{\mathbf{C}}^{1/2}$, and we observe that such matrices

217 are also symmetric, we can write, if we assume that the decomposition exists,
 218

$$\mathbf{C} = \mathbf{c}\mathbf{c} = \mathbf{c}^t\mathbf{c} = \mathbf{T}^t\hat{\mathbf{C}}\mathbf{T} = \mathbf{T}^t\hat{\mathbf{c}}^t\hat{\mathbf{c}}\mathbf{T} \quad (34)$$

219 implying that

$$\mathbf{c} = \hat{\mathbf{c}}\mathbf{T} \implies \mathbf{T} = \hat{\mathbf{c}}^{-1}\mathbf{c} \quad (35)$$

220 Even though such decomposition is valid for any choice of $\hat{\mathbf{C}}$, in practice
 221 it is important to choose such tensor *as close as possible* to its anisotropic
 222 counterpart, so to guarantee that for an initially isotropic material the ma-
 223 trix \mathbf{T} is exactly the identity. Following the ideas in [18] we choose $\hat{\mathbf{C}}$ so to
 224 minimize the Frobenius norm $\|\mathbf{C} - \hat{\mathbf{C}}\|_F$, with the additional constraint of ex-
 225 actly representing the bulk modulus defined in Eq. 7 of the original anisotropic
 226 tensor. This gives rise to the formulas

$$\hat{\mathbf{C}} = 3\left(\frac{\alpha}{3}\kappa\right)\mathbf{J} + 2\mu\mathbf{K} \quad (36)$$

227 where $\mathbf{J} := \mathbf{t}\mathbf{t}^t$ and $\mathbf{K} := \mathbf{I}_4 - \mathbf{J}$, with \mathbf{t} defined as

$$\mathbf{t} := \begin{pmatrix} \frac{1}{\sqrt{3}} \\ \frac{1}{\sqrt{3}} \\ \frac{1}{\sqrt{3}} \\ 0 \\ 0 \\ 0 \end{pmatrix} (3D), \quad \mathbf{t} := \begin{pmatrix} \frac{1}{\sqrt{2}} \\ \frac{1}{\sqrt{2}} \\ 0 \end{pmatrix} (2D) \quad (37)$$

228 and

$$\mathbf{I}_4 = \begin{pmatrix} 1 & 0 & 0 & 0 & 0 & 0 \\ 0 & 1 & 0 & 0 & 0 & 0 \\ 0 & 0 & 1 & 0 & 0 & 0 \\ 0 & 0 & 0 & 0.5 & 0 & 0 \\ 0 & 0 & 0 & 0 & 0.5 & 0 \\ 0 & 0 & 0 & 0 & 0 & 0.5 \end{pmatrix} (3D), \quad \mathbf{I}_4 = \begin{pmatrix} 1 & 0 & 0 \\ 0 & 1 & 0 \\ 0 & 0 & 0.5 \end{pmatrix} (2D) \quad (38)$$

229 Using Voigt's notation, the bulk modulus κ defined in Eq. 7 and appearing
 230 in Eq. 36 is

$$\kappa = \frac{\mathbf{m}^t\mathbf{C}\mathbf{m}}{\alpha^2} \quad (39)$$

231 a choice that enforces that *the bulk of the original anisotropic tensor \mathbf{C} co-*
 232 *incides exactly with the one of the "closest" tensor $\hat{\mathbf{C}}$.*

Under these assumptions, the 1st Lamé parameter μ of the closest isotropic tensor in Eq. 36 can be obtained in closed form from the minimization of the Frobenius error norm $\|\mathbf{C} - \hat{\mathbf{C}}\|_F$ to give

$$\mu = 0.2(C_{00} - 2C_{01} + C_{11} + C_{22}) \quad (2D) \quad (40a)$$

$$\begin{aligned} \mu = \frac{4}{33} [C_{00} - C_{01} - C_{02} + C_{11} - C_{12} + C_{22} \\ + \frac{3}{4}(C_{33} + C_{44} + C_{55})] \quad (3D) \end{aligned} \quad (40b)$$

233 3.2. Variational approach

With the proposed mapping, the mixed strain-displacement problem in Eq. 4a and Eq. 4b becomes

$$\int_{\Omega} \nabla^s \delta_{\mathbf{u}} : \mathbb{T}^t : \hat{\mathbb{C}} : \mathbb{T} : \boldsymbol{\varepsilon} = \int_{\Omega} \delta_{\mathbf{u}} \cdot \mathbf{f} \quad (41a)$$

$$- \int_{\Omega} \delta_{\boldsymbol{\varepsilon}} : \mathbb{T}^t : \hat{\mathbb{C}} : \mathbb{T} : (\boldsymbol{\varepsilon} - \nabla^s \mathbf{u}) = 0 \quad (41b)$$

which, once we define $\hat{\boldsymbol{\varepsilon}} := \mathbb{T} : \boldsymbol{\varepsilon}$ (and likewise for the test function) shows an obvious similarity with the isotropic case. The essential idea of our proposal is hence *to modify $\hat{\boldsymbol{\varepsilon}}$ instead of $\boldsymbol{\varepsilon}$* to obtain an equation in terms of the volumetric strain. Doing such exercise we obtain

$$\hat{\boldsymbol{\varepsilon}} = \mathbb{T} : \nabla^s \mathbf{u} - \frac{1}{\alpha} \text{Tr}(\mathbb{T} : \nabla^s \mathbf{u}) \mathbf{I} + \frac{1}{\alpha} \hat{\boldsymbol{\varepsilon}}^v \mathbf{I}$$

234 What follows is simply an algebraic exercise to follow the same steps as
235 in the general case, now particularised to the change of variables proposed.

236 Taking into account that $\mathbb{T}^{-1} : \mathbb{T} = \mathbb{T} : \mathbb{T}^{-1} = \mathbb{I}$ and that the trace can
237 be written as $\text{Tr}(\mathbb{T} : \nabla^s \mathbf{u}) = \mathbf{I} : \mathbb{T} : \nabla^s \mathbf{u}$, we obtain

$$\hat{\boldsymbol{\varepsilon}} = \mathbb{T} : \nabla^s \mathbf{u} - \frac{\mathbf{I} : \mathbb{T} : \nabla^s \mathbf{u}}{\alpha} \mathbb{T} : \mathbb{T}^{-1} : \mathbf{I} + \frac{\hat{\boldsymbol{\varepsilon}}^v}{\alpha} \mathbb{T} : \mathbb{T}^{-1} : \mathbf{I} \quad (42)$$

238 Premultiplying by $\mathbb{T}^{-1} :$ we can recover the enrichment of the original strain
239 as

$$\boldsymbol{\varepsilon} = \nabla^s \mathbf{u} - \frac{\mathbf{I} : \mathbb{T} : \nabla^s \mathbf{u}}{\alpha} \mathbb{T}^{-1} : \mathbf{I} + \frac{\hat{\boldsymbol{\varepsilon}}^v}{\alpha} \mathbb{T}^{-1} : \mathbf{I} \quad (43)$$

240 Note that for isotropic materials $\mathbb{T} = \mathbb{I}$, thus implying that the original
241 formulation is recovered.

Once arrived at this point, the derivation follows exactly the same path as in the general case. By substituting Eq. 42 into Eq. 41a and 41b we obtain

$$\int_{\Omega} \nabla^s \boldsymbol{\delta}_{\mathbf{u}} : \mathbb{T}^t : \hat{\mathbb{C}} : \left(\mathbb{T} : \nabla^s \mathbf{u} - \frac{\mathbf{I} : \mathbb{T} : \nabla^s \mathbf{u}}{\alpha} \mathbf{I} + \frac{\hat{\varepsilon}^v}{\alpha} \mathbf{I} \right) = \int_{\Omega} \boldsymbol{\delta}_{\mathbf{u}} \cdot \mathbf{f}$$

and, proceeding for the strain test function as for the strain,

$$- \int_{\Omega} \left(\nabla^s \boldsymbol{\delta}_{\mathbf{u}} : \mathbb{T}^t - \frac{\mathbf{I} : \mathbb{T} : \nabla^s \boldsymbol{\delta}_{\mathbf{u}}}{\alpha} \mathbf{I} + \frac{\delta_{\hat{\varepsilon}^v}}{\alpha} \mathbf{I} \right) : \hat{\mathbb{C}} : \left(-\frac{\mathbf{I} : \mathbb{T} : \nabla^s \mathbf{u}}{\alpha} \mathbf{I} + \frac{\hat{\varepsilon}^v}{\alpha} \mathbf{I} \right) = 0$$

Substituting $\mathbb{T}^t : \hat{\mathbb{C}} : \mathbb{T}$ by the original \mathbb{C} plus rearranging and collecting the relevant terms we then arrive to

$$\begin{aligned} & \int_{\Omega} \nabla^s \boldsymbol{\delta}_{\mathbf{u}} : \mathbb{C} : \nabla^s \mathbf{u} \\ & - \int_{\Omega} \left(-\frac{\mathbf{I} : \mathbb{T} : \nabla^s \boldsymbol{\delta}_{\mathbf{u}}}{\alpha} \right) \mathbf{I} : \mathbb{T}^{-t} : \mathbb{C} : \mathbb{T}^{-1} : \mathbf{I} \left(-\frac{\mathbf{I} : \mathbb{T} : \nabla^s \mathbf{u}}{\alpha} + \frac{\hat{\varepsilon}^v}{\alpha} \right) \\ & = \int_{\Omega} \boldsymbol{\delta}_{\mathbf{u}} \cdot \mathbf{f} \end{aligned} \quad (44a)$$

$$- \int_{\Omega} \left(\frac{\delta_{\hat{\varepsilon}^v}}{\alpha} \right) \mathbf{I} : \mathbb{T}^{-t} : \mathbb{C} : \mathbb{T}^{-1} : \mathbf{I} \left(-\frac{\mathbf{I} : \mathbb{T} : \nabla^s \mathbf{u}}{\alpha} + \frac{\hat{\varepsilon}^v}{\alpha} \right) = 0 \quad (44b)$$

242 Unfortunately, the previous form is not fully convenient for the mechan-
 243 ical response modelling as the constitutive law input strain would be $\nabla^s \mathbf{u}$
 244 rather than $\hat{\boldsymbol{\varepsilon}}$. This can be avoided by rearranging the enriched strain defi-
 245 nition in Eq. 42 as

$$\nabla^s \mathbf{u} = \mathbb{T}^{-1} : \hat{\boldsymbol{\varepsilon}} - \frac{1}{\alpha} \mathbb{T}^{-1} : \mathbf{I} (\hat{\varepsilon}^v - \mathbf{I} : \mathbb{T} : \nabla^s \mathbf{u}) \quad (45)$$

246 and substituting it into Eq. 44a.

We can now observe that with the proposed choice of “closest isotropic tensor” the equality

$$\hat{\kappa} := \frac{\mathbf{I} : \mathbb{T}^{-t} : \mathbb{C} : \mathbb{T}^{-1} : \mathbf{I}}{\alpha^2} = \frac{\mathbf{I} : \hat{\mathbb{C}} : \mathbf{I}}{\alpha^2} = \frac{\mathbf{I} : \mathbb{C} : \mathbf{I}}{\alpha^2} = \kappa$$

holds. This gives the final set of equations:

$$\int_{\Omega} \nabla^s \boldsymbol{\delta}_{\mathbf{u}} : \boldsymbol{\sigma}(\boldsymbol{\varepsilon}) - \int_{\Omega} (\mathbf{I} : \mathbb{T} : \nabla^s \boldsymbol{\delta}_{\mathbf{u}}) \hat{\kappa} (\hat{\varepsilon}^v - \mathbf{I} : \mathbb{T} : \nabla^s \mathbf{u}) = \int_{\Omega} \boldsymbol{\delta}_{\mathbf{u}} \cdot \mathbf{f} \quad (46a)$$

$$-\int_{\Omega} \delta_{\hat{\varepsilon}^v} \hat{\kappa} (-\mathbf{I} : \mathbb{T} : \nabla^s \mathbf{u} + \hat{\varepsilon}^v) = 0 \quad (46b)$$

In essence, the mixed formulation we propose in the anisotropic case consists in taking as unknowns the displacement \mathbf{u} and the modified volumetric strain

$$\hat{\varepsilon}^v = \mathbf{I} : \mathbb{T} : \nabla^s \mathbf{u} = \text{Tr}(\mathbb{T} : \nabla^s \mathbf{u})$$

247 instead of $\varepsilon^v = \text{Tr}(\nabla^s \mathbf{u}) = \nabla \cdot \mathbf{u}$.

248 3.3. Variational Multi-Scales stabilization

The discussion needs to be completed by the definition of a suitable stabilization. Proceeding similarly to the isotropic case, we can take a subgrid stabilization in the form (see Eqs. 17 with P_s the identity):

$$\mathbf{u}_s = \tau_1 [\mathbf{f} + \nabla \cdot (\mathbb{C} : \nabla^s \mathbf{u}_h + \hat{\kappa} (\hat{\varepsilon}_h^v - \mathbf{I} : \mathbb{T} : \nabla^s \mathbf{u}_h) \mathbf{I})] \quad (47a)$$

$$\hat{\varepsilon}_s^v = \tau_2 (\mathbf{I} : \mathbb{T} : \nabla^s \mathbf{u}_h - \hat{\varepsilon}_h^v) \quad (47b)$$

Upon substitution in the Galerkin form we obtain, when using linear elements (see Eq. 20):

$$\begin{aligned} & \int_{\Omega} \nabla^s \delta_{\mathbf{u}_h} : \boldsymbol{\sigma}(\boldsymbol{\varepsilon}_h) + \int_{\Omega} (1 - \tau_2) (\mathbf{I} : \mathbb{T} : \nabla^s \delta_{\mathbf{u}_h}) \hat{\kappa} (\hat{\varepsilon}_h^v - \mathbf{I} : \mathbb{T} : \nabla^s \mathbf{u}_h) \\ & = \int_{\Omega} \delta_{\mathbf{u}_h} \cdot \mathbf{f} \end{aligned} \quad (48a)$$

$$\begin{aligned} & \int_{\Omega} (1 - \tau_2) \delta_{\hat{\varepsilon}_h^v} \hat{\kappa} (\hat{\varepsilon}_h^v - \mathbf{I} : \mathbb{T} : \nabla^s \mathbf{u}_h) \\ & + \int_{\Omega} \tau_1 \hat{\kappa}^2 \nabla \delta_{\hat{\varepsilon}_h^v} \cdot \nabla \hat{\varepsilon}_h^v = - \int_{\Omega} \hat{\kappa} \nabla \delta_{\hat{\varepsilon}_h^v} \cdot \tau_1 \mathbf{f} \end{aligned} \quad (48b)$$

249 3.4. Finite Element Implementation - Anisotropic case

250 As for the isotropic case the residual in FEM notation assumes a slightly
251 different form depending on the choice of $\boldsymbol{\varepsilon}_h$. The option $\boldsymbol{\varepsilon}_h := \nabla^s \mathbf{u}_h - \frac{1}{\alpha} \nabla \cdot$
252 $\mathbf{u}_h \mathbf{I} + \frac{1}{\alpha} \hat{\varepsilon}_h^v \mathbf{I}$ gives

$$\mathbf{R}_I := \begin{pmatrix} N_I \mathbf{f}_{ext} - \mathbf{B}_I^t \boldsymbol{\sigma}(\boldsymbol{\varepsilon}_h) + \frac{1}{\alpha} \mathbf{B}_I^t \mathbf{C} \mathbf{T}^{-1} \mathbf{m} H_J - (1 - \tau_2) \hat{\kappa} \boldsymbol{\Psi}_I^t H_J \\ (1 - \tau_2) \hat{\kappa} N_I H_J + \hat{\kappa}^2 \mathbf{G}_I^t \tau_1 \mathbf{G}_J \hat{\varepsilon}_{h,J}^v - \hat{\kappa} \mathbf{G}_I^t \tau_1 \mathbf{f} \end{pmatrix} \quad (49)$$

253 while the choice $\boldsymbol{\varepsilon}_h := \nabla^s \mathbf{u}_h$ results in

$$\mathbf{R}_I := \begin{pmatrix} N_I \mathbf{f}_{ext} - \mathbf{B}_I^t \boldsymbol{\sigma}(\boldsymbol{\varepsilon}_h) - (1 - \tau_2) \hat{\kappa} \boldsymbol{\Psi}_I^t H_J \\ (1 - \tau_2) \hat{\kappa} N_I H_J + \hat{\kappa}^2 \mathbf{G}_I^t \tau_1 \mathbf{G}_J \hat{\varepsilon}_{h,J}^v - \hat{\kappa} \mathbf{G}_I^t \tau_1 \mathbf{f} \end{pmatrix} \quad (50)$$

254 with $\Psi_J := \mathbf{m}^t \mathbf{T} \mathbf{B}_J$,

$$H_J := N_J \varepsilon_{h,J}^v - \Psi_J^t \mathbf{u}_{h,J} \quad (51)$$

255 and

$$\hat{\kappa} := \frac{\mathbf{m}^t \mathbf{T}^{-t} \mathbf{C} \mathbf{T}^{-1} \mathbf{m}}{\alpha^2} \quad (52)$$

In either case, the LHS is identical and is given by

$$\text{LHS}_{IJ} := \begin{pmatrix} \mathbf{B}_I^t \mathbf{C} \mathbf{B}_J - (1 - \tau_2) \hat{\kappa} \Psi_I \Psi_J^t & (1 - \tau_2) \hat{\kappa} \Psi_I N_J \\ (1 - \tau_2) \hat{\kappa} N_I \Psi_J^t & - (1 - \tau_2) \hat{\kappa} N_I N_J - \hat{\kappa}^2 \mathbf{G}_I^t \tau_1 \mathbf{G}_J \end{pmatrix} \quad (53)$$

256 4. Results

257 4.1. Manufactured solution test

We begin the result section by verifying the convergence rates of the proposed formulation. To that end, we employ the Method of Manufactured Solutions [19] and focus on a problem defined over a unit square, positioned so that the bottom left corner coincides with the position (0,0). The chosen target displacement field is

$$\bar{\mathbf{u}} = A \begin{pmatrix} \sin(4\pi x) \\ \cos(4\pi y) \\ 0 \end{pmatrix}$$

where A represents an adjustable amplification factor which in our tests is set to 10^{-3} to ensure that the solution remains well within the small strain regime. Such displacement field yields the volumetric strain field

$$\bar{\varepsilon}^v = 4\pi A (\cos(4\pi x) - \sin(4\pi y))$$

The force field in equilibrium with such displacement can be obtained by substitution into Eq. 2 to give

$$\bar{\mathbf{f}} = (4\pi)^2 A \begin{pmatrix} C_{00} \sin(4\pi x) + C_{21} \cos(4\pi y) \\ C_{11} \cos(4\pi y) + C_{20} \sin(4\pi x) \\ 0 \end{pmatrix}$$

258 where the coefficients C_{ij} are the entries of the Voigt form of the constitu-
 259 tive tensor. For the sake of the benchmark, the domain is meshed using a
 260 linear quadrilateral structured mesh with 2^n lateral subdivisions. Different
 261 choices of the elastic parameters are employed with the aim of evaluating the
 262 performance in different conditions.

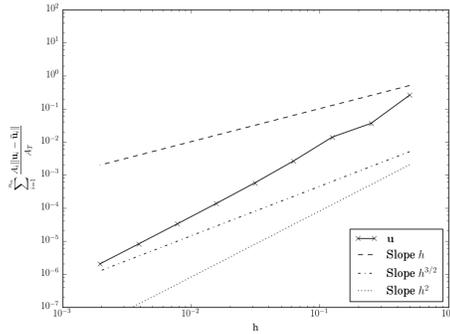
263 *4.1.1. Incompressible isotropic material*

264 A plain strain constitutive law with the material properties E and ν equal
 265 to 200 N/m^2 and 0.4999 is used with the aim of assessing the convergence at
 266 the incompressible limit.

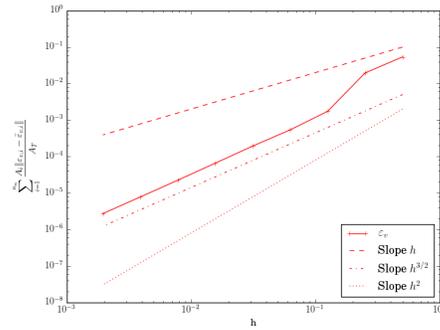
267 Table 1 collects the \mathbf{u} and ε^v error norms for each one of the meshes
 268 we use. These results are also depicted in Fig. 1. We observe that the
 269 convergence is quadratic for the \mathbf{u} field and $h^{3/2}$ for the ε^v field.

Table 1: Incompressible isotropic material manufactured solution test. \mathbf{u} and ε^v strain error norms.

n	1	2	3	4	5	6	7	8	9
h	0.5	0.25	0.125	6.25e-2	3.13e-2	1.56e-2	7.81e-3	3.91e-3	1.95e-3
$\ \mathbf{u} - \bar{\mathbf{u}}\ _{L^2(\Omega)}$	2.56e-1	3.57e-2	1.38e-2	2.60e-3	5.66e-4	1.34e-5	3.26e-5	8.06e-6	2.00e-6
$\ \varepsilon^v - \bar{\varepsilon}^v\ _{L^2(\Omega)}$	5.37e-2	1.95e-2	1.73e-3	5.37e-4	1.94e-4	6.56e-5	2.24e-5	7.79e-6	2.73e-6



(a) \mathbf{u} convergence.



(b) ε^v convergence.

Figure 1: Manufactured solution test. Incompressible isotropic material convergence analysis. τ_1 computed with \mathcal{C} .

270 *4.1.2. Anisotropic material*

271 A plane-strain anisotropic material is checked next, using the constitutive
 272 tensor

$$\mathbf{C} = \begin{pmatrix} 54469.29 & 8284.82 & 17726.94 \\ 8284.82 & 5981.77 & 2615.99 \\ 17726.94 & 2615.99 & 8305.89 \end{pmatrix} \quad (54)$$

273 The calculated \mathbf{C}_{iso} and \mathbf{T} matrices are

$$\mathbf{C}_{iso} = \begin{pmatrix} 29692.637 & 8817.713 & 0. \\ 8817.713 & 29692.637 & 0. \\ 0. & 0. & 10437.462 \end{pmatrix} \quad (55)$$

274 and

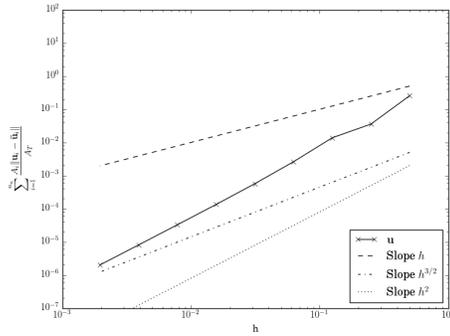
$$\mathbf{T} = \begin{pmatrix} 1.32161932 & 0.0931325 & 0.35389397 \\ -0.04531568 & 0.41023417 & -0.01086188 \\ 0.58737628 & 0.07153362 & 0.66756935 \end{pmatrix} \quad (56)$$

275 We recall that in the anisotropic case the obtained “volumetric strain” is
 276 not any longer $\nabla \cdot \mathbf{u}$ but instead $\mathbf{I} : \mathbb{T} : \nabla^s \mathbf{u}_h$. After computing the anisotropy
 277 matrix \mathbb{T} corresponding to the constitutive matrix in Eq. 54 we obtain the an-
 278 alytical volumetric strain field $\bar{\varepsilon}^v = 4\pi A (1.2763 \cos(4\pi x) - 0.503367 \sin(4\pi y))$.

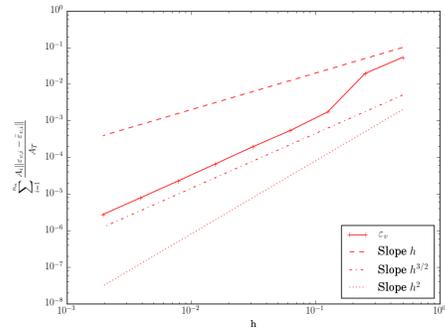
279 Table 2 collects the \mathbf{u} and $\hat{\varepsilon}^v$ error norms for each one of the meshes we
 280 use. These results are also depicted in Fig. 2.

Table 2: Anisotropic material manufactured solution test. \mathbf{u} and ε^v strain error norms.

n	1	2	3	4	5	6	7	8	9
h	0.5	0.25	0.125	6.25e-2	3.13e-2	1.56e-2	7.81e-3	3.91e-3	1.95e-3
$\ \mathbf{u} - \bar{\mathbf{u}}\ _{L^2(\Omega)}$	2.08e-2	1.94e-3	2.58e-3	1.33e-3	4.11e-4	1.09e-4	2.76e-5	6.92e-6	1.73e-6
$\ \hat{\varepsilon}^v - \bar{\varepsilon}^v\ _{L^2(\Omega)}$	1.94e-2	1.61e-2	2.13e-2	9.89e-3	3.03e-3	8.05e-4	2.07e-4	5.36e-5	1.41e-5



(a) \mathbf{u} convergence.



(b) \mathbf{mTBu} convergence.

Figure 2: Manufactured solution test. Anisotropic material convergence analysis. τ_1 computed with \mathbf{C} .

281 *4.2. Cook's membrane*

282 The second benchmark test considered is the well known Cook's mem-
 283 brane benchmark, described for example in [5]. The setup of the test is shown

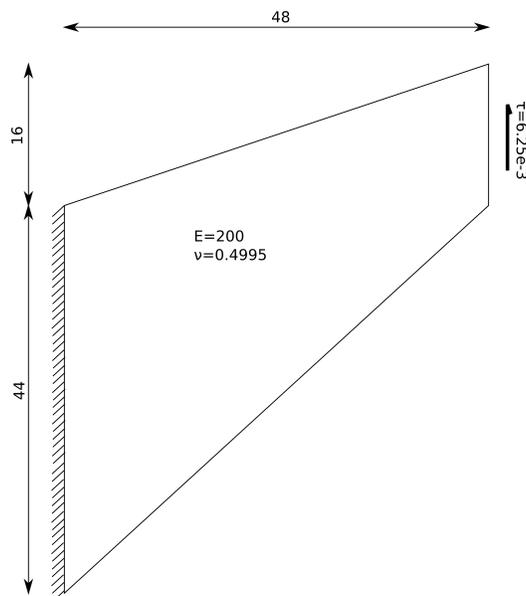


Figure 3: Setup of Cook's Membrane Benchmark [mm].

283 in Fig. 3. The test is run employing the proposed formulation using both
 284 triangles (irreducible and mixed) and quads (irreducible, mixed and Bbar).
 285 The test is also run in plane strain and in 3D.

287 *4.2.1. Incompressible isotropic material*

288 We firstly conduct the test using a linear elastic plane strain constitu-
 289 tive law with the material properties stated in Fig. 3. The plot of the
 290 y -displacement on the top right point is shown in Fig. 4 for a uniform
 291 mesh subdivisions by factors $2 \cdot 2^9$. We observe that the proposed formula-
 292 tion converges much faster to the expected value than the irreducible one.
 293 When comparing to the Q1P0 (Bbar) element, this shows a slightly better
 294 behaviour for the coarser meshes.

295 Complementarily, we solve the problem for a set of unstructured triangu-
 296 lar meshes whose sizes can be computed as $5/2^n$, $n \in (0, 6)$. Fig. 5 depicts
 297 the y -displacement convergence on the top right point. The superior perfor-
 298 mance of the mixed $\mathbf{u}-\varepsilon^v$ formulation becomes evident in this case.

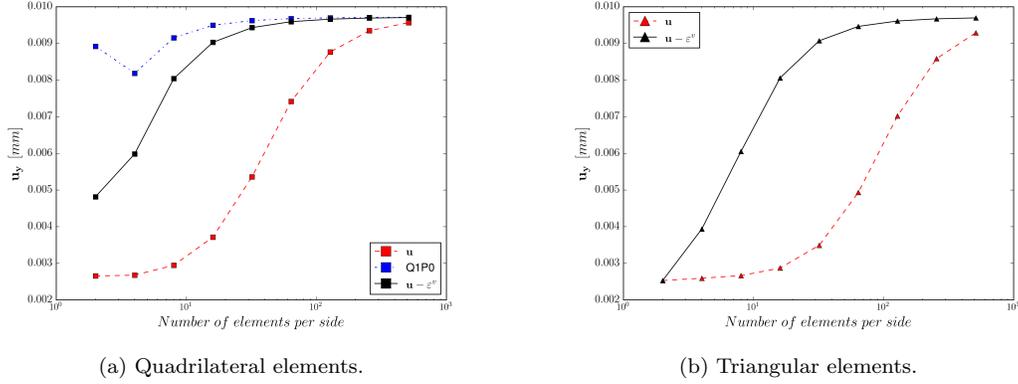


Figure 4: Cook's membrane test. Incompressible isotropic material u_y structured meshes convergence results.

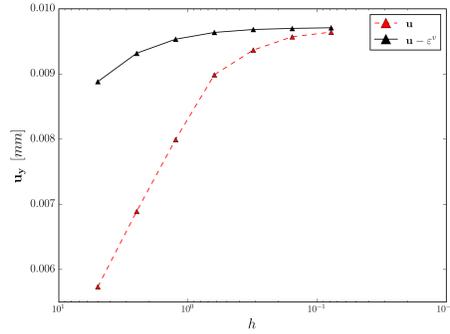


Figure 5: Cook's membrane test. Incompressible isotropic material u_y unstructured triangular mesh convergence results.

299 Finally, we also present a view of selected results in Fig. 6 showing that
 300 a good solution is found in all the variables of interest.

301 4.2.2. Incompressible anisotropic material

We carry out the same test but using an incompressible anisotropic material whose response is modelled by the constitutive tensor

$$\mathbb{C} = \begin{pmatrix} 970870.07 & 1239555.39 & 0.0 \\ 1239555.39 & 1622077.42 & 0.0 \\ 0.0 & 0.0 & 6711.41 \end{pmatrix}$$

302 with the associate \mathbf{C}_{iso} and \mathbf{T} matrices

$$\mathbf{C}_{iso} = \begin{pmatrix} 1292124.1915 & 1243904.9435 & 0. \\ 1243904.9435 & 1292124.1915 & 0. \\ 0. & 0. & 24109.624 \end{pmatrix} \quad (57)$$

303 and

$$\mathbf{T} = \begin{pmatrix} 0.34121548 & -0.20655167 & 0. \\ 0.53340404 & 1.31787552 & 0. \\ 0. & 0. & 0.52760836 \end{pmatrix} \quad (58)$$

304 Fig. 7 presents the convergence results. Once again, the proposed mixed
305 formulation far outperforms the irreducible approach.

306 4.3. Bimaterial Cook's membrane

307 4.3.1. Two isotropic materials

308 As a third test, we modify the second benchmark by introducing two
309 different materials as shown in Fig. 8. Only one of the two materials is
310 considered incompressible in order to introduce a large difference in the con-
311 stitutive behaviour. Thus, the E and ν are 2.0×10^4 Pa and 0.4995 in the
312 top half of the membrane while they are 2.0×10^2 and 0.3 in the bottom one.

313 We shall remark that introducing a discontinuity in the material is clas-
314 sically challenging for mixed approaches, but the proposed approach seems
315 to handle the case without difficulties, thus proving that one of the design
316 goals of the method is accomplished.

317 The plot of vertical displacement vs mesh subdivision for such configura-
318 tion is shown in Fig. 9.

319 Fig. 10 shows a view of the \mathbf{u} , ε^v and stress fields showing that no spurious
320 oscillations are found.

321 4.3.2. Isotropic - anisotropic materials

We repeat the same bimaterial Cook's membrane example but substi-
tuting the isotropic material in the bottom half of the membrane by the
anisotropic one characterized by the constitutive tensor

$$\mathbb{C} = \begin{pmatrix} 54469.29 & 8284.82 & 17726.94 \\ 8284.82 & 5981.77 & 2615.99 \\ 17726.94 & 2615.99 & 8305.89 \end{pmatrix}$$

322 The plot of vertical displacement vs mesh subdivision for such configura-
323 tion is shown in Fig. 11.

324 Fig. 12 shows a view of the \mathbf{u} , ε^v and stress fields showing that no spurious
 325 oscillations are found.

326 4.4. 3D anisotropic Cook's membrane

327 We extrude the same geometry 16 mm. The surface load is now $1.0 \times$
 328 10^5 N/mm². We fix the out of plane displacements in the front and rear
 329 surfaces. The anisotropic constitutive tensor we use is

$$\mathbf{C}_{aniso} := \begin{pmatrix} 5.99E+11 & 5.57E+11 & 5.34E+11 & 0 & 0 & 4.44E+09 \\ 5.57E+11 & 5.71E+11 & 5.34E+11 & 0 & 0 & -3.00E+09 \\ 5.34E+11 & 5.34E+11 & 5.37E+11 & 0 & 0 & 9.90E+05 \\ 0 & 0 & 0 & 1.92E+09 & 9.78E+06 & 0 \\ 0 & 0 & 0 & 9.78E+06 & 2.12E+09 & 0 \\ 4.44E+09 & -3.00E+09 & 9.90E+05 & 0 & 0 & 2.56E+10 \end{pmatrix} \quad (59)$$

330 We use an unstructured mesh conformed by around 230k linear tetrahe-
 331 dral elements (Fig 13).

332 As can be seen in Fig. 14 smooth results are obtained for all the fields
 333 thus confirming that the formulation also works correctly in the 3D case.

334 4.5. 3D necking bar

335 The objective of the benchmark is to compare the behaviour of the pro-
 336 posed formulation, using both a structured and unstructured discretization,
 337 to a reference Bbar implementation in a case involving plasticity. To that pur-
 338 pose we solve the well-known necking bar example using a perfect isotropic
 339 J2 plasticity law. The Young modulus, Poisson ratio, and yield stress are
 340 210×10^9 GPa, 0.29 and 200 MPa respectively. The specimen, whose dimen-
 341 sions are $5.4 \times 0.5 \times 0.2$ cm³, is clamped in its left face while an incremental
 342 total displacement of 0.006 cm is imposed in its right face.

343 A structured hexahedral mesh conformed by 4.4k elements (Fig. 15) is
 344 employed. A fairly similar discretization level in terms of element sizes is
 345 achieved with an unstructured mesh of around 33k tetrahedras (Fig. 16).

346 Figs. 17 and 18 present the plastic dissipation and the uniaxial stress ob-
 347 tained for the three cases. As it can be observed in Fig. 18 the final deformed
 348 shape and the uniaxial stress distribution is very similar in all the cases. No
 349 spurious oscillations are visible in the mixed solution. Plastic dissipation is
 350 slightly underestimated in the unstructured mesh results, probably because
 351 of a slightly stiffer behaviour of the tetrahedral element.

352 *4.6. Automotive machinery piece*

353 This last example presents the (purely qualitative) results of a simulation
354 involving the plastic deformation of an industrial piece. The problem con-
355 sists in the mechanical analysis of an aluminium object from the automotive
356 industry. The specimen (Fig. 19) has a length around 280 mm and a thick-
357 ness of 8.5 mm. It is clamped in the magenta region in Fig. 19c. A surface
358 load of 300 kPa is incrementally applied in the yellow region in Fig. 19c.

359 The material response is modeled using an isotropic small strain perfect
360 J2 plasticity law. E and ν are set to 70 GPa and 0.35. The plastic regime is
361 characterized by the yield stress $\sigma_y = 120$ MPa. Such material model implies
362 a quasi-incompressible behaviour within the plastic region (implying that
363 the volumetric deformation will be small compared to the total deformation),
364 thus making unappealing the use of low order irreducible elements. The com-
365 plexity of the shape prevents the use of B-bar type hexahedral meshes, thus
366 leaving the proposed $\mathbf{u}\text{-}\varepsilon^v$ technology as one of the few possible alternatives.

367 The domain was meshed using 550k linear tetrahedra, employing the
368 proposed mixed formulation.

369 Figs. 20 and 21 depict the obtained results. The piece shows a rather
370 ductile behaviour up to the point at which a plastic hinge appears in the
371 vicinity of the clamping (Fig. 21c).

372 Fig. 21 collects a set of snapshots describing the evolution of the plastic
373 deformation. More specifically, it can be noted that prior to the formation
374 of the plastic hinge at the basis, large parts of the specimen reach the yield
375 stress (120 MPa) (Figs. 21a and 21b) and thus present a plastic energy
376 dissipation (Fig. 21c).

377 **5. Conclusion**

378 The paper presents a novel mixed element, able to tackle the quasi-
379 incompressible limit. The proposed formulation aims at addressing problems
380 with material nonlinearity, and is effective also in presence of multiple mate-
381 rial interfaces. A convenient modification, that allows dealing with initially
382 anisotropic problems is described. The proposed mixed method is proved
383 effective in combination with a plastic material behaviour.

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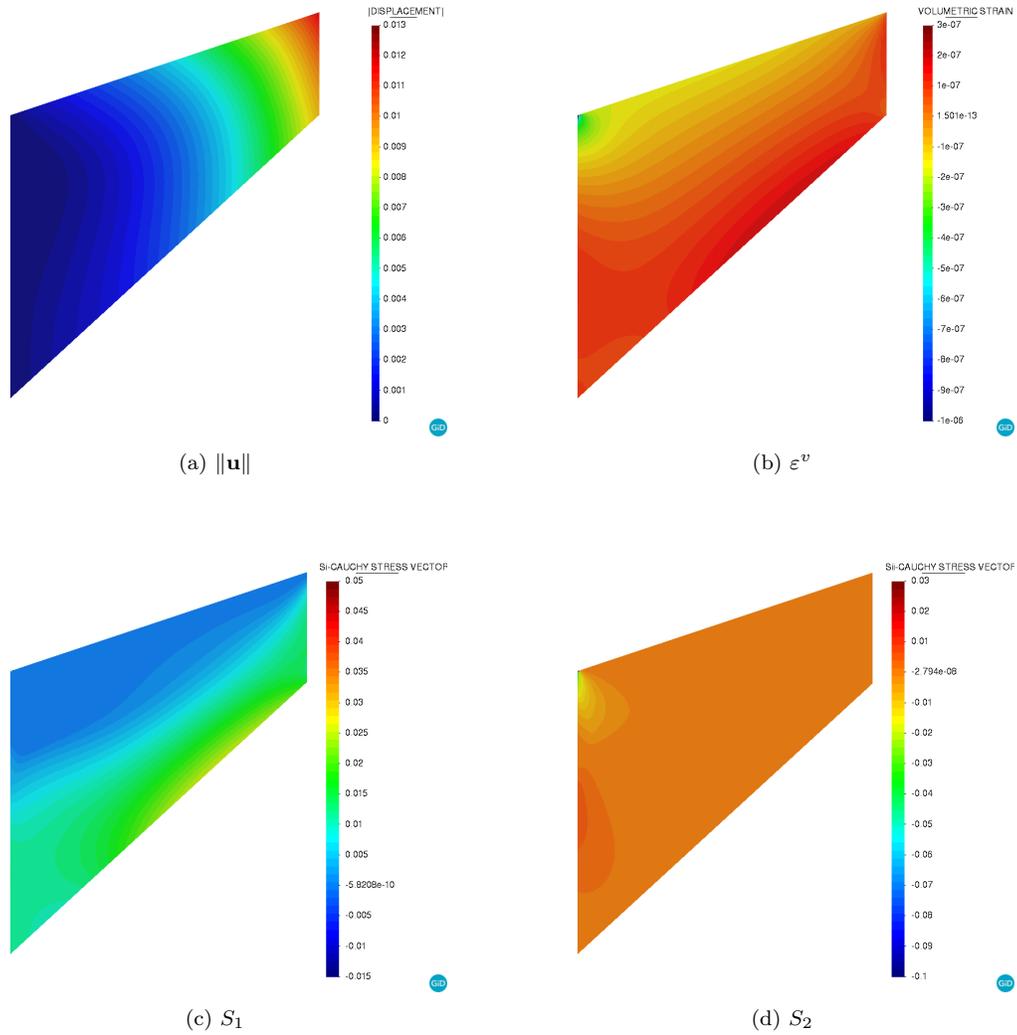


Figure 6: Cook's membrane test. Solution snapshots for the 256 divisions quadrilateral mesh (S_1, S_2 : principal stresses).

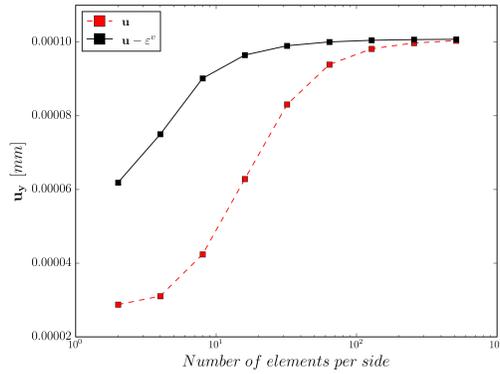


Figure 7: Cook's membrane test. Incompressible anisotropic material u_y convergence results.

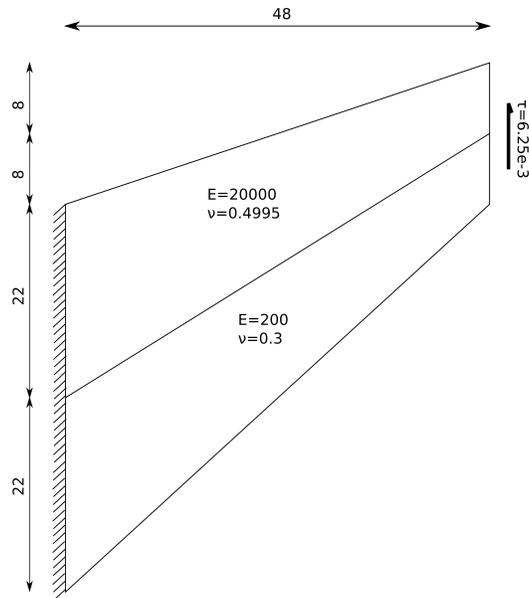


Figure 8: Setup of Cook's Membrane Benchmark using two distinct materials.

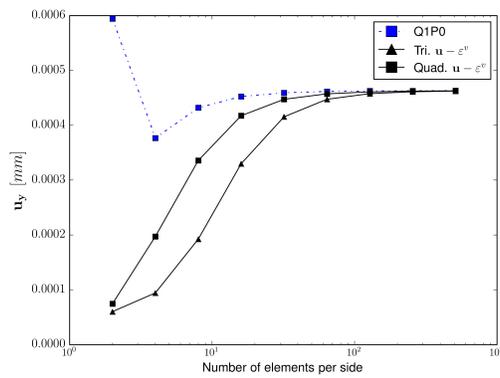


Figure 9: Bimaterial Cook's membrane test. u_y convergence.

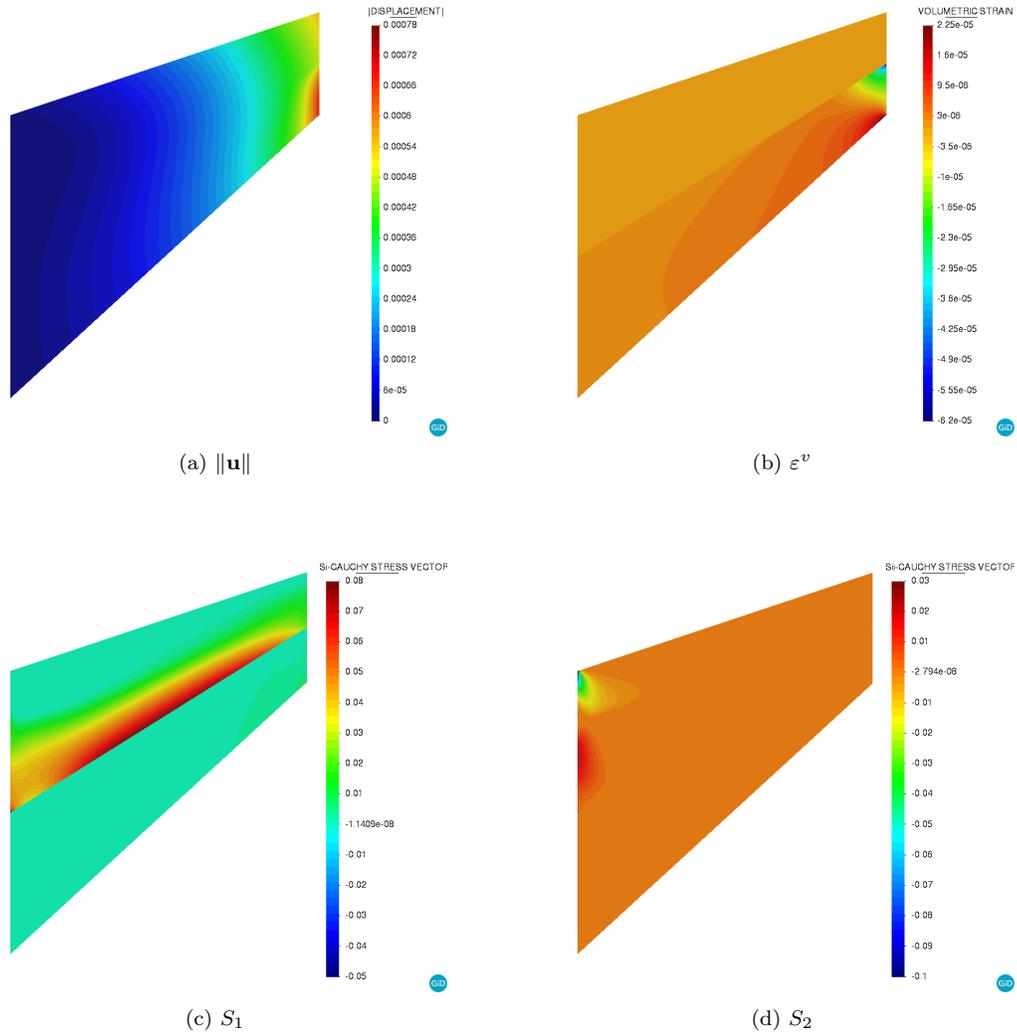


Figure 10: Bimaterial Cook's membrane test. Solution snapshots for the 256 divisions quadrilateral mesh.

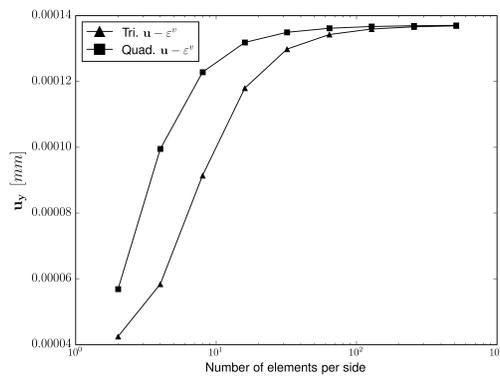


Figure 11: Bimaterial (isotropic - anisotropic) Cook's membrane test. u_y convergence.

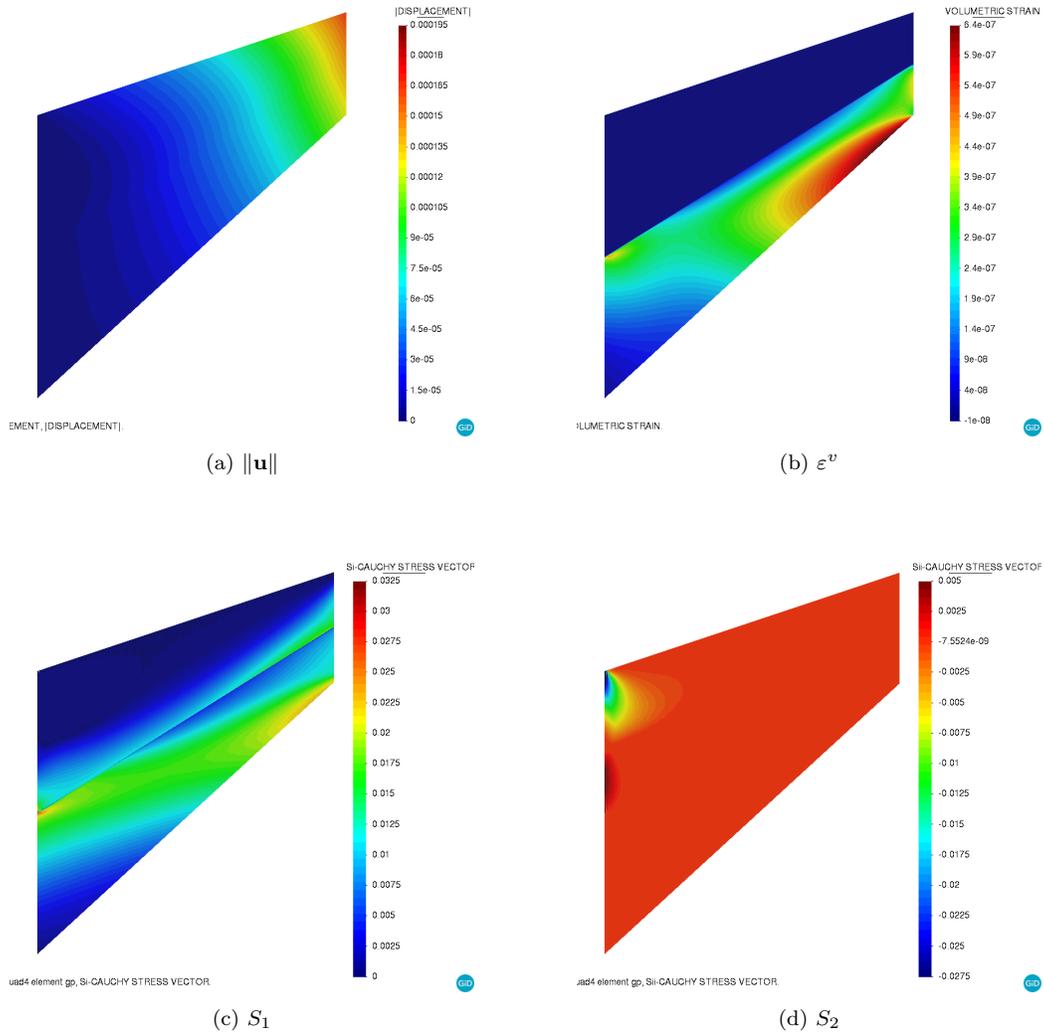


Figure 12: Bimaterial (isotropic - anisotropic) Cook's membrane test. Solution snapshots for the 256 divisions quadrilateral mesh.

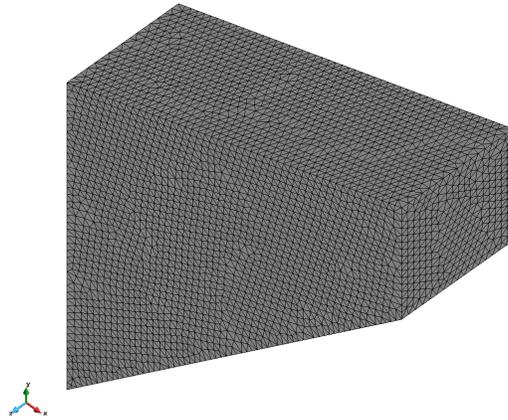


Figure 13: 3D anisotropic Cook's membrane. Unstructured linear tetrahedra mesh.

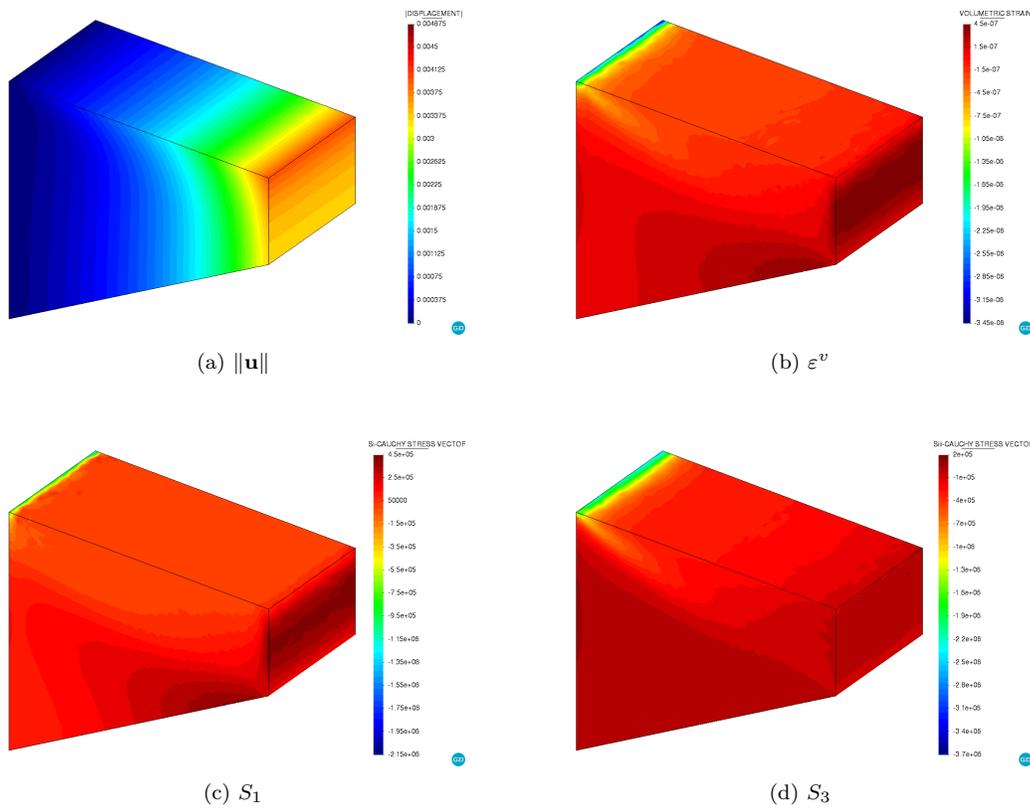
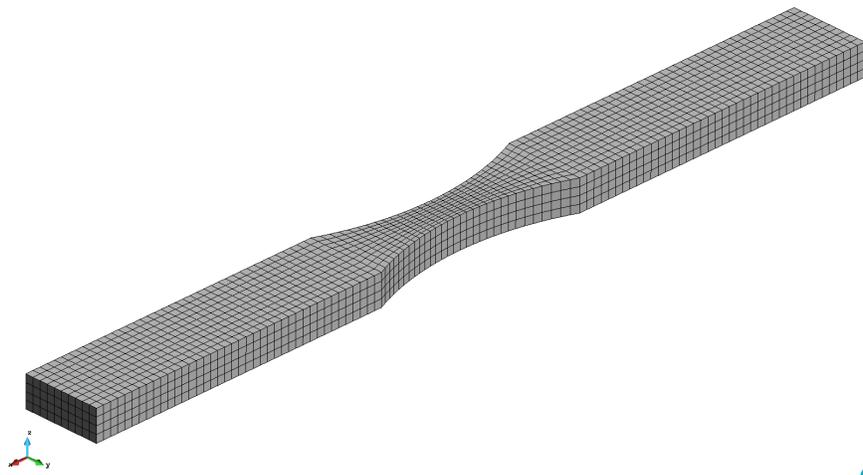
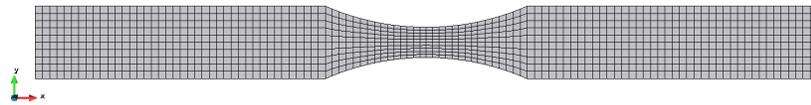


Figure 14: 3D anisotropic Cook's membrane. Solution snapshots.

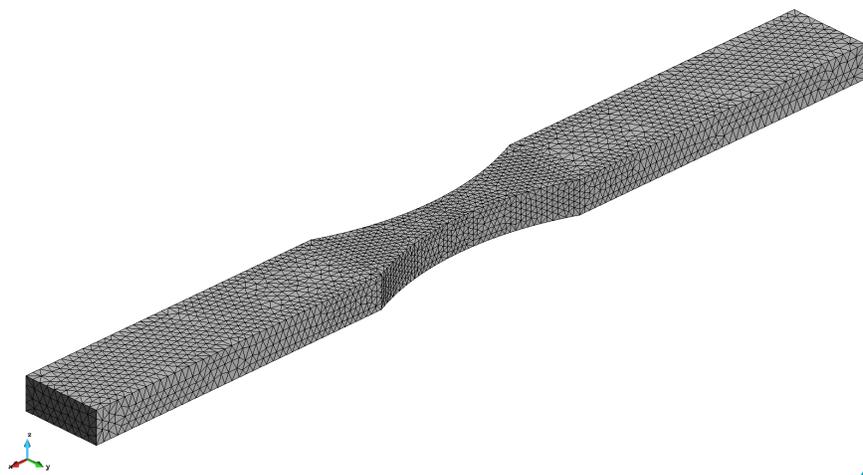


(a) Isometric view.

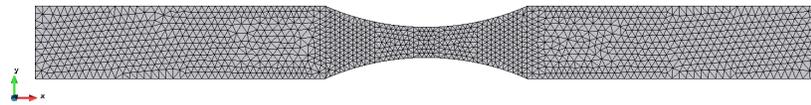


(b) xy -plane view.

Figure 15: 3D necking bar. Structured hexaedral mesh.



(a) Isometric view.



(b) xy -plane view.

Figure 16: 3D necking bar. Structured tetrahedra mesh.

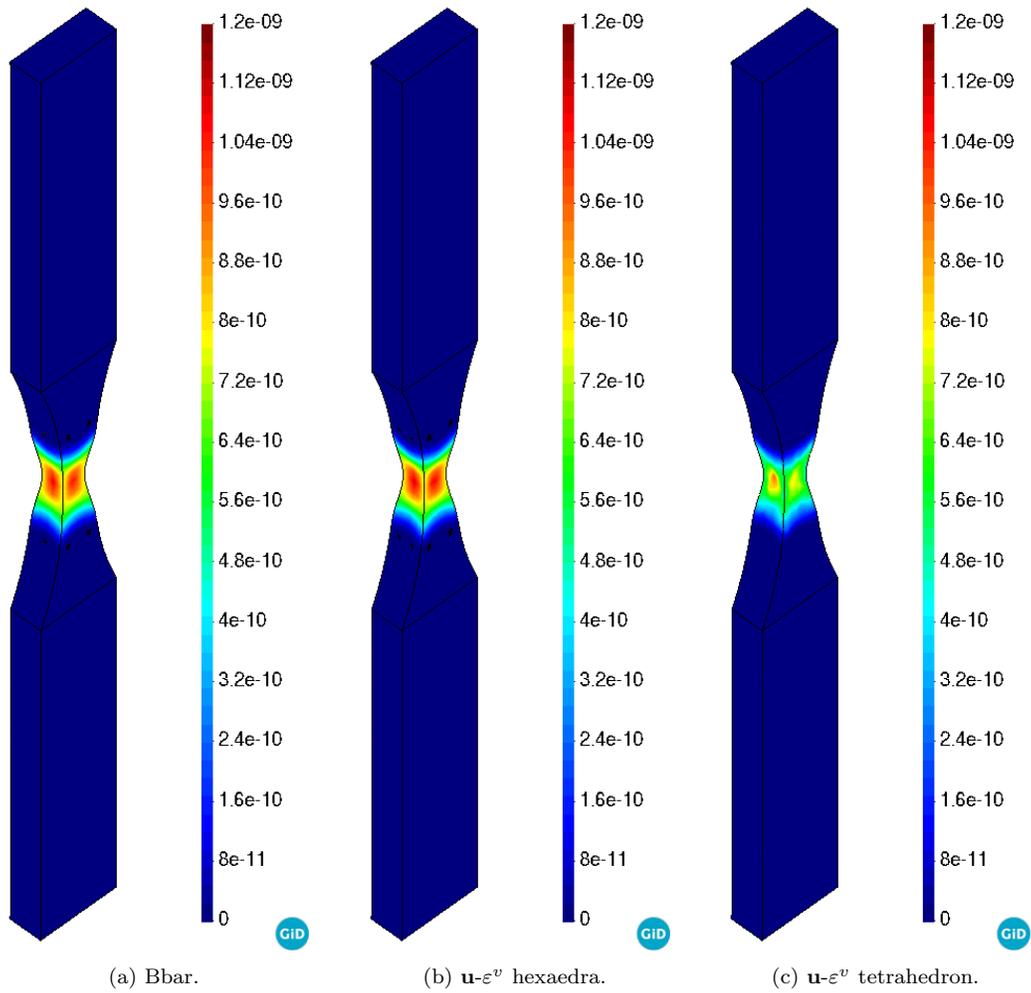
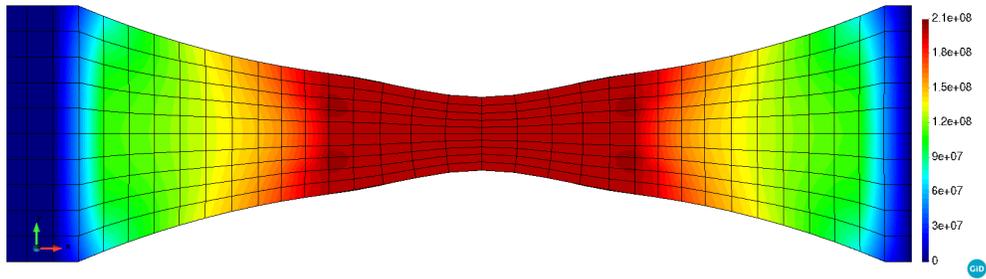
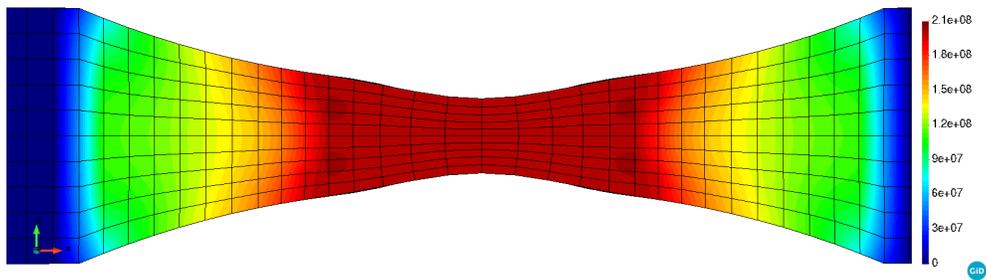


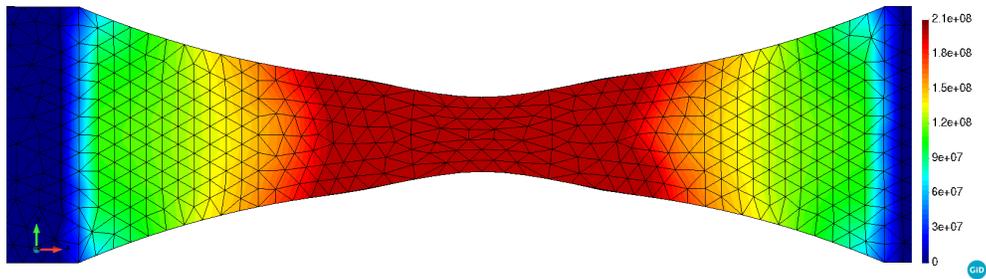
Figure 17: 3D necking bar. Plastic dissipation (deformation scale x40).



(a) Bbar.

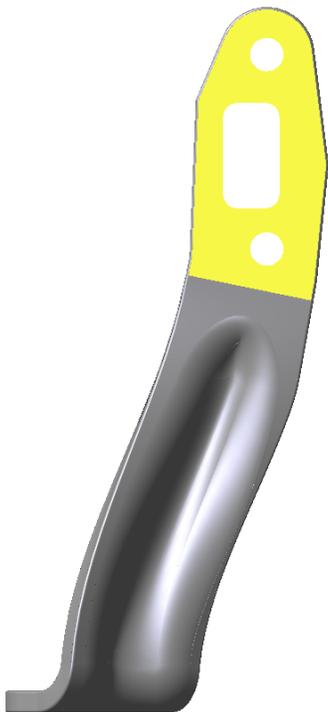


(b) $\mathbf{u}-\varepsilon^v$ hexaedra.



(c) $\mathbf{u}-\varepsilon^v$ tetrahedron.

Figure 18: 3D necking bar. Uniaxial stress [Pa] (deformation scale x40).



(a) *yz*-plane view.



(b) *y*-axis isometric view 1.



(c) *y*-axis isometric view 2.

Figure 19: Automotive machinery piece. Problem geometry.

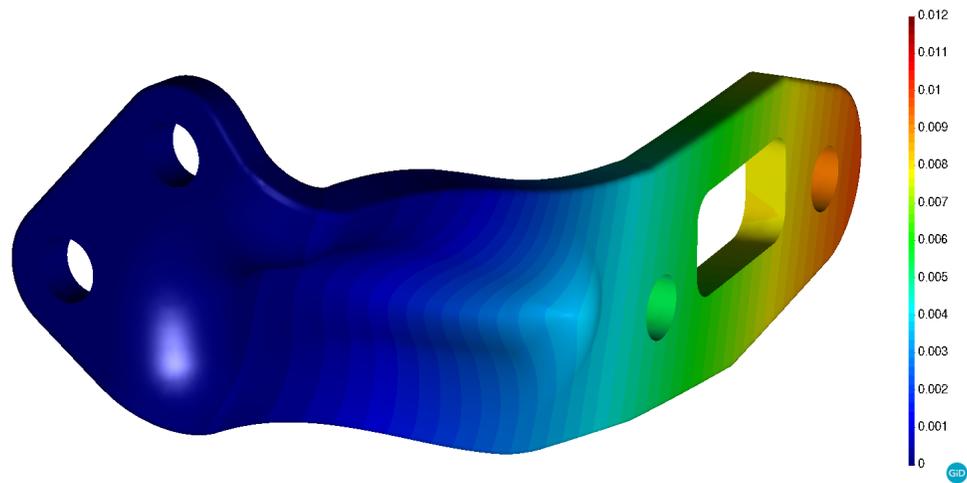
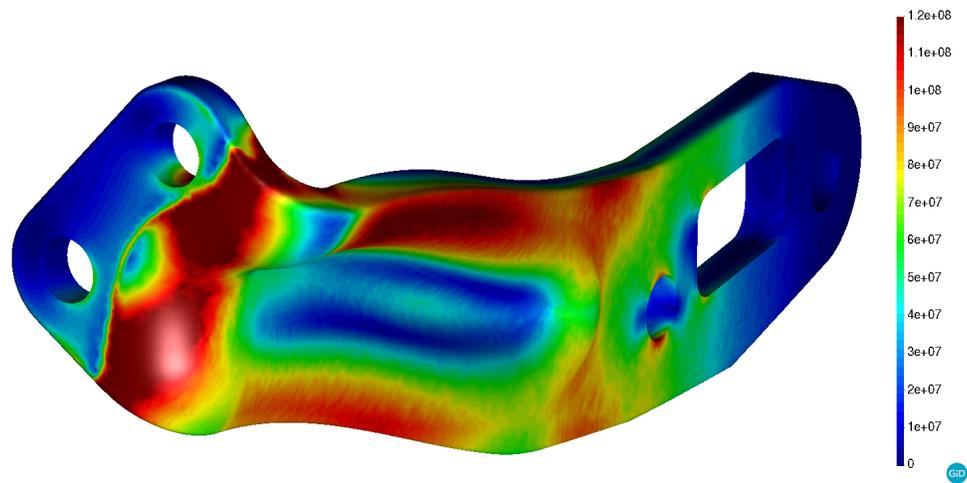
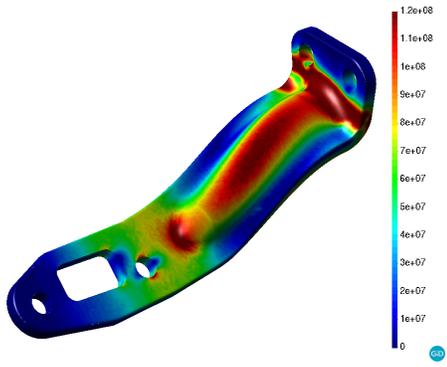


Figure 20: Automotive machinery piece. $\|\mathbf{u}\|$ [m].



(a) Uniaxial stress rear view [Pa] (deformation scale $\times 5$).



(b) Uniaxial stress lateral view [Pa].



(c) Plastic dissipation (log scale).

Figure 21: Automotive machinery piece. Plasticity magnitudes isometric view.